

# Fabrication and Characterization of Hematite/Titanium Dioxide Photocatalyst for Hydrogen Production

Weerapong Bootluck

A Thesis Submitted in Fulfillment of the Requirements for the Degree of Doctor of Philosophy in Sustainable Energy Management Prince of Songkla University

2022

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ชื่อวิทยานิพนธ์	การสังเคราะห์และศึกษาคุณสมบัติตัวเร่งปฏิกิริยาเชิงแสงชนิด
	เฮมาไทด์/ไทเทเนียมไดออกไซด์สำหรับผลิตไฮโดรเจน
ผู้เขียน	นายวีรพงค์ บุตรลักษณ์
สาขา	การจัดการพลังงานอย่างยั่งยืน
ปีการศึกษา	2564

## บทคัดย่อ

การศึกษานี้มีวัตถุประสงค์เพื่อพัฒนาตัวเร่งปฏิกิริยาเชิงแสงเพื่อใช้สำหรับผลิตแก๊สไฮโดรเจน โดยใช้วัสดุผสมชนิดไทเทเนียมไดออกไซด์กับเฮมาไทด์ ร่วมกับแพลลาเดียม และการปรับปรุงผิวด้วย เทคนิคพลาสมา ผลจากการศึกษาการปรับปรุงตัวเร่งปฏิกิริยาเชิงแสงชนิด α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd พบว่า ลักษณะโครงสร้างสันฐานวิทยามีขนาดโครงสร้างผลึกเฉลี่ยเท่ากับ 22 นาโนเมตร และมีแถบช่องว่าง พลังงานเท่ากับ 2.00 อิเล็กตรอนโวล โดยแถบช่องว่างพลังงานน้อยกว่าชนิดไททาเนียมไดออกไซด์ บริสุทธิ์ (3.16 อิเล็กตรอนโวลต์) ขนาดพื้นที่ผิวของตัวเร่งปฏิกิริยาเพิ่มขึ้นและมีรัศมีความต้านทาน EIS น้อยส่งผลทำให้เกิดการเคลื่อนที่ระหว่างประจุได้ดี โดยตัวเร่งปฏิกิริยาเซิงแสงชนิด α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd มีประสิทธิภาพในการผลิตไฮโดรเจนได้ดีที่สุด เท่ากับ 3490.54 มิลลิโมลต่อชั่วโมง และให้ประสิทธิภาพในการนำกลับมาใช้ซ้ำในการเร่งปฏิกิริยาเชิงแสงสำหรับผลิตไฮโดรเจนได้เป็น อย่างดี

การศึกษาการปรับปรุงผิวตัวเร่งปฏิกิริยาเชิงแสงชนิด α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> ด้วยเทคนิคพลาสมา จากการศึกษาพบว่าลักษณะโครงสร้างสันฐานวิทยา พบว่าตัวเร่งปฏิกิริยาชนิด α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10 มี ความเหมาะสมที่สุด โดยผลจากการศึกษาโครงสร้างผลึกไททาเนียมไดออกไซด์โครงสร้างผลึก เปลี่ยนเป็นโครงสร้างผลึกผสมแบบอะนาเทสและแบบรูไทล์ โดยมีขนาดอนุภาคโครงสร้างผลึกเฉลี่ย ลดลงจาก 25 ถึง 20 นาโนเมตร แถบช่องว่างพลังงานลดลงจาก 2.18 ถึง 1.84 อิเล็กตรอนโวลต์ และ สามารถดูดกลืนแสงในช่วงวิสิเบิล (ช่วงความยาวคลื่น 400-700 นาโนเมตร) จากนั้นศึกษาความ ต้านทานทางไฟฟ้าเคมีพบว่าลดลงจาก 2.24 ถึง 1.90 กิโลโอห์ม การปรับปรุงผิวด้วยอาร์กอน พลาสมายังส่งผลให้เกิดช่องว่างของออกซิเจนและTi<sup>3+</sup> ซึ่งส่งผลให้ประสิทธิภาพการเร่งปฏิกิริยาเชิง แสงสำหรับผลิตไฮโดรเจนภายใต้ช่วงแสงวิสิเบิลได้ดี เมื่อนำตัวเร่งปฏิกิริยาเชิงแสงนำไปทดสอบการ แยกแก๊สไฮโดรเจนในช่วงแสงวิสิเบิล พบว่าตัวเร่งปฏิกิริยาเชิงแสงชนิด α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10 สามารถ ผลิตแก๊สไฮโดรเจนดีที่สุดเท่ากับ 1.25 มิลลิโมลต่อชั่วโมง คำนำ : โฟแตแคตะลิส เฮมาไทด์ ไทเทเนียมไดออกไซด์ ไฮโดรเจน

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### ABSTRACT

This research aims to the development of photocatalytic  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> for hydrogen production by using noble metal deposition and plasma treatment. This study was to modification of heterojunctions of hematite ( $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>) deposition on titanium dioxide (TiO<sub>2</sub>) and Palladium (Pd) co-catalyst. As the result of the characterization, the average crystallite size of ternary  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd was 22 nm and its band gap energy was 2.00 eV, much lower than that of the pure TiO<sub>2</sub> nanoparticles (3.16 eV). The  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd also has higher specific surface area and smaller EIS radius, which enhance interface activity and charge transfer. The  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd exhibited great performance, with H<sub>2</sub> production rate of 3490.54 µmol h<sup>-1</sup> g<sup>-1</sup> and excellent stability in multi-cycle H<sub>2</sub> production.

Plasma modification on α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> photocatalyst, the crystalline phase of TiO<sub>2</sub> form changed from pure anatase to a mixture of anatase and rutile phases. The particle size was reduced from 25 to 20 nm. The band gap energy decreased from approximately 2.18 eV to 1.84 eV and exhibited a wider absorption edge under visible light irradiation (400–700 nm). The charge transfer resistance was reduced from 2.24 to 1.90 kΩ. Ar-plasma modification of the surface showed the attribution of oxygen vacancies and Ti<sup>3+</sup>. The plasma treatment improved the heterojunction photocatalyst to enhance the photocatalytic activity under visible light for H<sub>2</sub> evolution. The best photocatalytic activity for H<sub>2</sub> evolution rate was obtained at 1.25 mmol h<sup>-1</sup> for α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10 photocatalyst

Keyword : Photocatalyst, Hematite, Titanium Dioxide, Hydrogen

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## LIST OF ABBREVIATION

Α	Constant
α	The absorbance
Ag/AgCl	Reference Electrode
AFM	Atomic force microscopy
Ar	Argon
β	The peak width at half maximum (FWHM) of the XRD peak
Cu <sub>2</sub> O	Copper (I) oxide
CoFe <sub>2</sub> O <sub>4</sub>	Cobalt ferrite
d	The crystallite size (nm)
θ	The diffraction angle
Eg	Band gap energy
EIS	Electrochemical impedance
EtOH	Ethanol
Fe(NO <sub>3</sub> ) <sub>3</sub> .9H <sub>2</sub> O	Iron(III) nitrate nonahydrate
FTO	Fluorine-doped tin oxide
Hz	Hertz
ICP-OES	Inductively coupled plasma optical emission spectrometer
k	Constant related to the crystallite shape normally taken as 0.89
a-Fe <sub>2</sub> O <sub>3</sub>	Hematite
GC	Gas chromatography
GO	Graphite oxide
g-C <sub>3</sub> N <sub>4</sub>	Graphitic carbon nitrides
h	Planck's constant
λ	The X-ray wavelength (nm)
mV	Millivolts
nm	Nanometer
Na <sub>2</sub> SO <sub>4</sub>	Sodium sulfate

# LIST OF ABBREVIATION (CONTINUED)

Pd	Palladium
PL	Photoluminescence spectroscopy
Pt	Platinum
UV-Vis	UV-vis spectrometry
SEM	Scanning electron microscopy
TCD	Thermal conductivity detector
Ti (OC(CH <sub>3</sub> ) <sub>3</sub> ) <sub>4</sub>	Titanium (IV) butoxide
Ti	Titanium foil
TiO <sub>2</sub>	Titanium dioxide
TEM	Transmission electron microscopy
XPS	X-ray photoelectron spectroscopy
SnO <sub>2</sub>	Tin (IV) oxide
ν	The light frequency
W	Watts
WO <sub>3</sub>	Tungsten trioxide
XRD	X-ray Diffractometry
ZnO	Zinc Oxide

## **LIST OF PAPERS**

I. Bootluck, W., Chittrakarn, T., Techato, K., Jutaporn, P and Khongnakorn, W. 2021. S-Scheme  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> Photocatalyst with Pd Cocatalyst for Enhanced Photocatalytic H<sub>2</sub> Production Activity and Stability. Catalysis Letters. 1-17.

II. Bootluck, W., Chittrakarn, T., Techato, K and Khongnakorn, W. 2021. Modification of surface  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> photocatalyst nanocomposite with enhanced photocatalytic activity by Ar gas plasma treatment for hydrogen evolution. Journal of Environmental Chemical Engineering, 9 (4), 105660.

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# CHAPTER 1 INTRODUCTION

## 1.1 Background

The energy crisis of the fossil fuel shortage and the increase of pollutant is the main issues of global concern. Nowadays, fossil fuels are increasing in energy demand [1,2]. Hydrogen (H<sub>2</sub>) is considered one of the most promising renewable energy for sustainable energy and the environment. H<sub>2</sub> is an energy-efficient, non-pollutant and clean energy (H<sub>2</sub> burned only by-product is water) [3] Hydrogen production from solar light includes photoelectrochemical, photobiological and photocatalytic, the method is low cost and environmental friendly [4,5].

Photocatalytic water splitting for hydrogen production is one of the most methods to produce H<sub>2</sub> from solar, photocatalytic methods is a promising technology for H<sub>2</sub> evolution and separation that provides an application in H<sub>2</sub> evolution and wastewater treatment[6]. A semiconductor is a catalytic material that activity with a photon for hydrogen evolution and degradation of pollutants. Among semiconductor materials that have been used for photocatalytic efficiencies such as cadmium sulfide (CdS) [7], Zinc oxide (ZnO)[8], tungsten trioxide (WO<sub>3</sub>) [9] and titanium dioxide (TiO<sub>2</sub>) [10]. TiO<sub>2</sub> is the most widely used semiconductor material based for the photocatalytic process. It is used for a semiconductor in photocatalytic due to chemical stability, high reusability, low cost and excellent photocatalytic activity which include wide band gap energy (3.2 eV) [10]. Because of the wide bandgap energy of TiO<sub>2</sub> is absorption from the visible light which activity of photocatalytic. However, TiO<sub>2</sub> is limited for absorption from the solar region (visible light of 43 %) because of the pure TiO<sub>2</sub> absorption only of UV light radiation about 4 % of the total light [10,11]. The methods to modify the surface of semiconductor material are a key factor for photocatalytic efficiency. Numerous methods have been modifying photocatalytic material such as noble metal doping, metal doping [12], iondoping [13] and modification surface [10]. Hematite ( $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>) is a noble metal oxide, with wider absorption visible light (400-700 nm) and low band gap energy (2.0-2.3 eV), due to its excellent nontoxic and is cost-effective [10].

The noble metal deposition of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> on the TiO<sub>2</sub> can improve to wider absorption under visible light. Moreover, the deposition of ternary on the heterojunction can be improved photocatalytic activity, trapping the recombination electron-hole pair. The enhanced efficiency of photocatalytic activity has been reported deposition on heterojunction photocatalyst by platinum (Pt), silver (Ag) and palladium (Pd) etc. Palladium (Pd) is one of the noble metals due to its chemical properties are similar to platinum (Pt), high surface area, and great electron transfer chemical stability [11]. For the modification surface, the plasma treatment is a method for modifying the surface by etching of surface which changes the surface properties to enhanced photocatalytic activity. In addition, these methods of modifying to generating of oxygen vacancy and highly stable of Ti<sup>3+</sup> on the surface of TiO<sub>2</sub>, the increasing of the surface area including absorption under visible light and UV radiation and improve photogeneration of charge carrier [10].

This research aimed to improve the photocatalytic material of heterojunction of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> with the ternary of Pd nanoparticle, and modification of the surface by plasma treatment for obtain for improve photocatalytic activity for hydrogen production under solar light and visible light. The photocatalytic membrane was to modify by plasma for H<sub>2</sub> production. The characterization of material in the surface morphology, chemical crystallite size, charge carrier separation, charge transfer, particle size, optical bandgap energy, elemental composition and chemical state, respectively. Finally, the photocatalytic material was investigated for H<sub>2</sub> production, stability and investigated in H<sub>2</sub> production.

## 1.2 The objective of this research was as follows

- To synthesis of the heterojunction hematite  $(\alpha$ -Fe<sub>2</sub>O<sub>3</sub>) on titanium dioxide (TiO<sub>2</sub>) photocatalytic and cocatalyst with palladium (Pd)
- To modified of photocatalytic hematite (α-Fe<sub>2</sub>O<sub>3</sub>) on titanium dioxide (TiO<sub>2</sub>) photocatalytic by plasma treatment for hydrogen evolution

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### **CHAPTER 2**

This study presents the development of heterojunction S-Scheme  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> photocatalyst with Pd cocatalyst for enhanced photocatalytic H<sub>2</sub> production activity and stability. The  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd was obtained characterization by various techniques including XRD, UV-vis, FE-SEM, TEM, EDS and XPS, respectively. The migration of photogenerated charge carriers was performed by PL emission and transient photocurrent response. The ternary heterojunction of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd exhibits improve photocatalytic activity towards H<sub>2</sub> evolution rate and its stability under exposure to visible light is investigated. This article has been published in the title of "S-Scheme  $\alpha$ -Fe <sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> Photocatalyst with Pd Cocatalyst for Enhanced Photocatalytic H<sub>2</sub> Production Activity and Stability in Catalysis Letters.

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S-Scheme  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> Photocatalyst with Pd Cocatalyst for Enhanced Photocatalytic H<sub>2</sub> Production Activity and Stability.

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# S-Scheme $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> Photocatalyst with Pd Cocatalyst for Enhanced Photocatalytic H<sub>2</sub> Production Activity and Stability

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### Abstract

In this work, hematite with titanium dioxide ( $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>) heterojunctions with palladium (Pd) nanoparticles were synthesised to improve efficiency of photocatalytic hydrogen production.  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> was loaded onto TiO<sub>2</sub> surfaces, then Pd nanoparticles were deposited to make a ternary photocatalyst. The chemical composition, morphology and surface properties of photocatalytic ternary heterojunction were characterized by XRD, UV–Vis, FE-SEM, TEM, EDS, XPS techniques and BET analysis. The PL emission, transient photocatalyst nanocomposites. The average crystallite size of ternary  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd was 22 nm and its band gap energy was 2.00 eV, much lower than that of the pure TiO<sub>2</sub> nanoparticles (3.16 eV). The  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd also has higher specific surface area and smaller EIS radius, which enhance interface activity and charge transfer. The  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd exhibited great performance, with H<sub>2</sub> production rate of 3490.54 µmol h<sup>-1</sup> g<sup>-1</sup> and excellent stability in multi-cycle H<sub>2</sub> production. The photocatalytic mechanism of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd as explained by the S-scheme heterojunction is that the electron in the VB of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and TiO<sub>2</sub> are transferred to CB of each photocatalyst. Then, the electrons in the CB of TiO<sub>2</sub> are transferred to the VB of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and the photogenerated electrons in CB of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> can migrate to Pd, which increase the redox ability for H<sub>2</sub> production and increase the separation of photogenerated e<sup>-</sup>-h<sup>+</sup> pairs. Overall, the experimental results and theoretical analyses confirm the high potential for the applicability of the ternary photocatalysts for H<sub>2</sub> production.

**Graphical Abstract** 



Keywords Photocatalysis  $\cdot$  Heterojunction  $\cdot \alpha$ -Fe<sub>2</sub>O<sub>3/</sub>TiO<sub>2</sub>-Pd  $\cdot$  Hydrogen production  $\cdot$  Pd-ternary

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1 Introduction

Developments in clean hydrogen power has risen rapidly during the last decade. The global hydrogen market is forecasted to increase to \$155 billion in 2022, and the annual growth rate is higher than 6% [1]. Currently, industrial hydrogen production is mainly through steam reforming using fossil fuels and natural gas [2, 3]. However, fossil fuels use in hydrogen production is not considered sustainable for the future [4, 5]. Alternative renewable energy sources for hydrogen production include solar, wave, and biomass energies [6]. Solar energy could be converted to hydrogen through photoelectrolytic [7], photoelectrochemical [8, 9], photocatalytic [10], and photobiological methods [11]. Particularly, the photocatalytic approach is a promising technology as it is environment-friendly, technically feasible, and economically favourable [12], especially with photocatalytic water splitting [10, 12, 13]. Recent researches on photocatalytic water splitting has focused on improving hydrogen production rate and widening wavelength range of usable solar light [14, 15].

Several semiconductor materials, such as cadmium sulfide (CdS), zinc oxide (ZnO), tungsten trioxide (WO<sub>2</sub>), bismuth vanadate (BiVO<sub>4</sub>), iron (III) oxide (Fe<sub>2</sub>O<sub>3</sub>), molybdenum disulfide (MoS<sub>2</sub>) graphitic carbon nitride (g-C<sub>3</sub>N<sub>4</sub>) nanocomposites [9, 10] and titanium dioxide  $(TiO_2)$  can be used as photocatalytic materials [14–16]. Among these, TiO<sub>2</sub> is often used in photocatalytic water splitting owing to its good reusability, chemical stability, economic feasibility, and low environmental impact [17, 18]. The key factors determining the performance of a photocatalyst include structure and morphology, crystallinity, band gap energy  $(E_{\sigma})$ , and corrosion resistance. The high band gap energy of TiO2 (3.2 eV) means it requires UV light to excite the electrons from valence band (VB) to the conduction band (CB), and UV radiation makes up just about 4% of the total energy in solar radiation [19, 20]. However, the efficiency of photocatalytic H<sub>2</sub> production over TiO<sub>2</sub> remains low because the photogenerated electrons in the CB quickly recombine with the holes (h<sup>+</sup>) in the VB [21, 22]. Hence, to utilize the visible portion of sunlight (43%,  $\lambda$  = 380–750 nm), it is necessary to improve the band gap energy of TiO2 [23, 24]. Many attempts have been made to hinder the e<sup>-</sup>-h<sup>+</sup> recombination. For instance, methods such as doping, noble metal deposition, non-metal ions, and surface sensitization [9] have been used to modify photocatalytic TiO2-based materials, to reduce the recombination of e<sup>-</sup>-h<sup>+</sup> pairs and to advance the response of the photocatalyst to visible light [25, 26]. In photocatalytic heterojunctions, the direct contact of two semiconductors with suitable CB and VB potentials can provide charge transfer through the interface. Various

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types of heterojunctions formed in TiO<sub>2</sub>-based composites have been shown to increase charge lifetime and enhance photocatalytic activity, such as WO<sub>3</sub>/TiO<sub>2</sub>, Cds/TiO<sub>2</sub>, ZnO/ TiO<sub>2</sub>, Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>, etc. [27, 28].

The  $Fe_2O_3/TiO_2$  nanocomposite photocatalyst has been synthesized with heterojunction at the interface to improve photoactivity for the removal of pollutants from wastewater, and for water splitting in hydrogen production. Iron oxide  $(Fe_2O_2)$  is a metal oxide that can have various crystalline structures, such as magnetite (Fe<sub>3</sub>O<sub>4</sub>), maghemite ( $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>), hematite ( $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>), etc. [29]. Hematite ( $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>) is the most stable form of iron oxides and it also has a narrow range of band gap energy of 2.0-2.3 eV [30-32]. Therefore, it can absorb light up to 600 nm and can also collect as much as 40% of the energy in solar radiation, based on the spectral energy distribution. Recently, α-Fe<sub>2</sub>O<sub>3</sub> has been used as a photocatalytic material for hydrogen production via water splitting [33-35] due to its outstanding advantages, including low cost, chemical stability, and non-toxicity [19, 36]. However, Fe2O3 also suffers from a high e-h+ recombination rate. Its poor conductivity means that an overpotential is needed to produce H<sub>2</sub>. For example, Zhu et al. [37] and Huang et al. [38] developed two types of semiconducting heterojunctions of α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> to enhance photoactivity under visible light irradiation. Furthermore, Zhu et al. [37] has synthesized a hierarchical structure (H-Fe<sub>2</sub>O<sub>2</sub>/TiO<sub>2</sub>). which showed higher efficiency in photocatalytic H<sub>2</sub> production under visible light. Madhumitha et al. [33] prepared α-Fe2O3/TiO2 core-shell nanoparticle photocatalysts by coprecipitation and the obtained heterojunction could decrease charge transfer from TiO<sub>2</sub> to  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> by suppressing e<sup>-</sup>-h<sup>+</sup> pair recombination.

On the other hand, noble metals deposition on semiconducting heterojunctions can improve the efficiency of photoelectron generation while inhibiting e<sup>-</sup>-h<sup>+</sup> recombination, resulting in faster H<sub>2</sub> production. The enhanced efficiency with a ternary heterojunction photocatalyst at the interface of noble metal has been reported [27-29]. The ternary material nanocomposite was synthesized to eliminate the drawback of photocatalytic materials because the Schottky barrier formation and plasmonic Z-scheme at the heterojunction photocatalyst interface can prevent electron-hole recombination [27-29]. Güy et al. [16] introduced the ternary MoS2/Ag/Ag3VO4 to delay recombination of photocarriers by plasmonic Z-scheme structure formation. Atacan et al. [8] prepared Ag2CrO4/GO/ MnFe2O4 composite that was able to enhance H2 production and increase electron density with Z-scheme structure. In addition, the step-scheme (S-scheme) heterojunction photocatalyst was divided into two types of semiconductor photocatalyst (p-type or n-type). S-scheme formed the photogenerated transfer of charge carriers and promoted the separation electron-hole pair. The internal electric

field is the main driving force of charge transfer between two semiconductor photocatalysts [39, 40]. Recently, the S-scheme ternary heterojunction has been reported [41, 42]. Zhu et al. [41] have studied the effectiveness of NiO as a co-catalyst of g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> on S-scheme photocatalyst heterojunction, which showed high-efficiency in the prevention of electrons and hole recombination. Hu et al. [42] developed a ternary S-scheme of Ni<sub>2</sub>P and SnNb<sub>2</sub>O<sub>6</sub>/ CdS-diethylenetriamine (SNO/CdS-D) to ease charge separation and high hydrogen harvesting. Yan et al. [43] studied the S-scheme ternary photocatalyst heterojunction of  $Co_3S_4/g$ – $C_3N_4/\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. They showed high stable of hydrogen evolution due to high adsorption rate and high redox potential. Hence, the co-catalyst process is a promising technology to enhance photocatalyst properties. Palladium (Pd) is one of the noble metals that has been deposited on heterojunctions, and its chemical properties are similar to platinum (Pt). The advantages offered by Pd include low cost, stability, high surface area, and great electron transfer [44, 45]. The deposition of Pd nanoparticles can modify the photocatalytic surface and crystalline structure of the heterojunction and enhance the separation of e<sup>-</sup>-h<sup>+</sup> pairs. The Pd co-catalyst could increase the photocatalytic hydrogen production rate by prolonging the lifetime of photogenerated e<sup>-</sup>-h<sup>+</sup> pairs. In addition, Ramírez et al. [46], reported an increase in charge carriers with Pd nanoparticles deposited on WO<sub>3</sub>-TiO<sub>2</sub> heterojunction. When Pt. Pd. and silver (Ag) were used as co-catalysts. Pd increased the photocatalytic activity and enabled strong recycling of the photocatalyzed reaction [47]. Chuaicham et al. [48] reported that the deposition of ternary Pd over sepiolight/g-C<sub>3</sub>N<sub>4</sub> resulted in high efficiency of photocatalysis due to visible light absorption and prevention of e<sup>-</sup>-h<sup>+</sup> recombination at the heterojunction of photocatalyst nanocomposite.

Given that deposition of Pd over WO<sub>3</sub>-TiO<sub>2</sub> and sepiolight/g-C<sub>3</sub>N<sub>4</sub> heterojunctions could increase the photocatalytic activity and prevent e<sup>-</sup>-h<sup>+</sup> recombination, a ternary photocatalyst S-scheme of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> with ternary Pd heterojunctions would be able to combine the advantages of this nanocomposite photocatalyst with an improvement in stability of H<sub>2</sub> production. Based on the literature review above, the performance of ternary Pd nanoparticles deposition on photocatalytic α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> heterojunction for H<sub>2</sub> generation have not been evaluated. Hence, in this study, ternary Pd is introduced into heterojunction photocatalyst nanocomposites, to form ternary heterojunction  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd nanocomposites. These nanocomposites morphology and pore structure are characterized by various techniques include FE-SEM, TEM, EDS, XRD, UV-Vis, XPS, and BET respectively. The migration photogenerated of charge carriers was performed by PL emission, EIS and transient photocurrent response. The ternary heterojunction

of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd exhibit improve photocatalytic activity towards H<sub>2</sub> evolution rate and its stability under exposure to visible light are investigated.

### 2 Materials and Methods

### 2.1 Materials

Titanium butoxide  $(Ti(OC(CH_3)_3)_4)$  and palladium chloride  $(PdCl_2)$  were obtained from Sigma-Aldrich. Ethylene glycol (EG) was purchased from Loba Chemie. Iron (III) nitrate nonahydrate (Fe(NO<sub>3</sub>)<sub>3</sub>·9H<sub>2</sub>O) was procured from Ajax Finechem. Ethanol and methanol (AR, 99.9%) were purchased from RCI Labscan.

### 2.2 Synthesis of TiO<sub>2</sub> Nanoparticles

 $TiO_2$  was prepared using a method previously reported by Peng et al. [49] with slight modifications. First, 3 mL of titanium butoxide and 30 mL of ethylene glycol were added into a flask and heated at 180 °C for 2 h. After the solution was allowed to cool down to room temperature, the white precipitate products were centrifuged. The precipitates were then washed with ethanol several times, followed by drying at room temperature. Finally, the products were calcined at 450 °C for 2 h, then TiO<sub>2</sub> was obtained as dried powder.

### 2.3 Preparation of $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>

To prepare  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>, 1.5 g of TiO<sub>2</sub> powder was added in 30 mL ethanol with 50 wt% Fe (NO<sub>3</sub>)<sub>3</sub>·9H<sub>2</sub>O. The selected synthesis proportions, with 50 wt% loading of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> on TiO<sub>2</sub> nanocomposite, gave the best photocatalytic activity as reported by Huang et al. [38]. The sample was stirred for 30 min and then evaporated at 50 °C. The photocatalytic Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> nanocomposite was then calcined at 300 °C for 10 min. Finally, the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> nanocomposite was heated at 5 °C/min to reach 300 °C and calcined for 6 h and the product was labelled as  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>.

### 2.4 Pd Deposition on α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> Nanoparticles

The  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> nanoparticles were dispersed in 50 mL deionized (DI) water. Then, PdCl<sub>2</sub> solution was dropped into the dispersion under magnetic stirring for a 1.0 wt% loading following a procedure reported by Chuaicham et al. [48], and irradiated under a 250 W Hg arc lamp for 1 h [50]. Afterwards, the sample was washed with ethanol several times and dried at 60 °C for 120 min. The prepared sample was labelled as  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd. The preparation procedure is schematically shown in Fig. 1.



Fig.1 Schematic illustration of the synthesis of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd photocatalytic heterojunction nanocomposite

### 2.5 Material Characterisation

Surface morphology of the synthesised  $\alpha$ -Fe<sub>2</sub>O<sub>2</sub>/TiO<sub>2</sub>-Pd nanocomposite were analysed using field-emission scanning electron microscopy (FESEM, Apreo, FEI, USA) and transmission electron microscopy (TEM; Tecnai G2 F20). The textural properties of the photocatalysts was analyzed for their porosity, specific surface area (SSA) and nitrogen (N<sub>2</sub>) adsorption-desorption isotherm by Brunauer-Emmett-Teller (BET) analyzer (Micromeritics ASAP 2460.USA) [51]. Before BET analyses, the temperature was set at 190 °C for 3 h for degassing [52]. The pore size distribution was calculated from the desorption curve by BJH method. The average crystallite size was determined using an X-ray diffractometer (XRD; X'Pert MPD, PHILIPS, Netherlands) by scanning of 20 range from 20° to 80°. The calculation of crystallite size and crystallinity followed Scherrer's equation [53]:

$$D = \frac{\kappa \lambda}{\beta \text{Cos}\theta} \tag{1}$$

where *k* is a shape factor constant (0.9),  $\lambda$  is the X-ray wavelength (nm),  $\theta$  is the diffraction angle, *D* is the crystallite size (nm), and  $\beta$  is the full width at half maximum (FWHM) of the diffraction line. Electronic states of elements on the heterojunction surface were analysed by X-ray photoelectron spectroscopy (XPS; AXIS Ultra DLD, Krotos, UK). The band gap energy was calculated from the UV–Vis spectrum recorded with a UV-2401 instrument (Shimadzu, Japan). The recombination and separation efficiency of charge carriers was investigated with a photoluminescence spectrometer (PL; Avantes AvaSpec-2048, Netherlands). The transient photocurrent, electrochemical impedance spectroscopy (EIS) and Mott-Schottky measurements were conducted with a potentiostat (Potentiostat/Galvanostats PGSTAT302N

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Metrohm, Autolab, Netherland). The instrument configuration was set according to a protocol published previously [42–44] and the working electrode was prepared following Duo et al. [54].

The average crystallite size was calculated from the XRD peak using the Debye–Scherrer equation [53, 55]. The band gap energy near valence band of TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles was calculated from the following Tac's plot equation [56, 57]:

$$\alpha hv = A(hv - E_g)^{n/2} \tag{2}$$

where  $\alpha$  is an absorption coefficient, *h* is a Planck constant, *v* is an incident light frequency, *A* is a constant,  $E_g$  is the band gap and *n* is an integer. The value of *n* depends on the type of optical transition and the value of 1 was selected for direct observed optical transition for all the samples [56]. The VB of TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> was determined by using XPS spectra. The CB and band edge position were performed according to our previous study [57]. The work function of photocatalyst was calculated according to Ren et al. [58].

### 2.6 Photocatalytic Activity of H<sub>2</sub> Production

The photocatalytic reactor was set up with a plexiglas cell of 100 mL. A 300 W UV lamp and visible light mercury lamps (250 W,  $\lambda \ge 420$  nm) were used as light sources. The  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd photocatalyst (0.1 g) was added to 100 mL solution of a sacrificial agent (methanol: water = 30:70 v/v). Before the reaction, nitrogen gas was delivered into the reactor for 60 min [4]. The produced H<sub>2</sub> was analysed by gas chromatography (GC, Hewlett Packard, HP 6890, USA) with a thermal conductivity detector (TCD) and using helium as the carrier gas. The generated H<sub>2</sub> amount was calculated by Heider and Kang method [59]. For multi-cycle H<sub>2</sub> production, after 4-h cycle, the used catalyst was thoroughly cleaned and dried then reused again for the next cycle. The recycling stability of the sample was calculated following the protocol according to a previous study [60].

### **3** Results and Discussion

### 3.1 Characterisation of α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd Photocatalytic Nanocomposites

The XRD patterns of different types of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd synthesized are shown in Fig. 2. There are five characteristic diffraction peaks of pure anatase TiO<sub>2</sub> at 2θ = 25.35°, 37.84°, 48.14°, 53.97°, and 55.18°. The hematite phase of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (JCPDS 73-2234) contains peaks at 2θ = 24.16°, 33.18°, 35.65°, 40.88°, 49.49°, 54.10°, 62.47°, and 64.04°, for which can be attributed to (012), (104), (101), (113),

S-Scheme  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> Photocatalyst with Pd Cocatalyst for Enhanced Photocatalytic...



Fig.2 XRD patterns of pure  $\alpha\mbox{-}Fe_2O_3,\mbox{Ti}O_2,\mbox{Ti}O_2\mbox{-}Pd,\mbox{}\alpha\mbox{-}Fe_2O_3/\mbox{Ti}O_2$  and  $\alpha\mbox{-}Fe_2O_3/\mbox{Ti}O_2\mbox{-}Pd$ 

 $\ensuremath{\text{Table 1}}$  The average crystallite size, crystallinity, and band gap energy of various samples

Sample	Average crystallite size (nm)	Crystallinity (%)	Band gap energy ( $E_{g;}$ eV)
TiO <sub>2</sub>	$19.01 \pm 4.95$	95.01	3.16
$\alpha$ -Fe <sub>2</sub> O <sub>3</sub>	$21.78 \pm 4.36$	65.82	2.02
TiO <sub>2</sub> -Pd	$20.66 \pm 2.02$	95.54	3.14
α- Fe <sub>2</sub> O <sub>3</sub> /TiO <sub>2</sub>	$23.50 \pm 6.65$	92.55	2.18
α-Fe <sub>2</sub> O <sub>3</sub> /TiO <sub>2</sub> -Pd	$22.11 \pm 3.56$	94.54	2.00

(024), (116), (214), and (300), respectively. The XRD patterns of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd samples confirm that they contain hematite phase of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and anatase phase of TiO<sub>2</sub>, which agrees with a previous study [38].The XRD pattern of Pd nanoparticles shows a peak at 2 $\theta$ =39.98° representing a lattice plane (111). The loading of Pd nanoparticles did not display any change in XRD peaks, possibly because of the small particle size and/or the low loading level on the TiO<sub>2</sub>, as also reported in previous studies [50, 61]. However, the presence of Pd doping could be confirmed by XPS spectrum, which will be further discussed in the following section.

Table 1 shows the average crystallite size, the crystallinity, and band gap energy of various samples. The reported crystallite size is an average from replicate measurement. The average crystallite size of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> photocatalyst was slightly increased from those of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and TiO<sub>2</sub> nanoparticles [49]. Loading Pd on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> resulted in smaller crystallite size, which helps increasing surface area for photocatalytic activity [45]. The crystallinities of pure TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> were 95.01% and 65.82%, respectively, while crystallinity was 92.55% for  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> sample. After loading Pd nanoparticles onto the heterojunction  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> surfaces, the average crystallinity was increased to 94.54%. The high crystallinity is expected to enhance the photocatalytic activity by improving the separation of photoinduced e<sup>-</sup>–h<sup>+</sup> pairs [23, 62].

The band gap energy  $(E_g)$  was reduced when  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and Pd were added onto the TiO<sub>2</sub> surface, as shown in Table 1.  $E_g = 2.18$  and 2.00 eV were recorded for  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd, respectively. The decreasing of  $E_g$  of TiO<sub>2</sub> (3.16 eV) to 2.00 eV in  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd indicates the decrease in e<sup>-</sup>-h<sup>+</sup> recombination rate under solar light.

The UV-Vis DSR data of TiO<sub>2</sub>, TiO<sub>2</sub>-Pd, and α-Fe<sub>2</sub>O<sub>3</sub>/ TiO<sub>2</sub>-Pd are shown in Fig. 3a. Pure TiO<sub>2</sub> and TiO<sub>2</sub>-Pd showed absorption only in the UV range (<400 nm) at energies exceeding their band gap energy (3.16 and 3.14 eV, respectively, Table 1). On the other hand,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> showed strong light absorption over the visible range ( $\lambda > 420$  nm). The  $\alpha$ -Fe<sub>2</sub>O<sub>2</sub>/TiO<sub>2</sub> heterostructures displayed significantly enhanced absorption in the visible region from 400 to 800 nm, and the high absorption window expanded when α-Fe<sub>2</sub>O<sub>3</sub> was loaded onto the TiO<sub>2</sub> nanoparticles. In the heterojunction α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> composites, the visible light absorption is caused by  $2({}^{6}A_{1}) \rightarrow ({}^{4}T_{1})$  ligand field transitions of Fe<sup>3+</sup> [38].Compared to heterojunction  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/ TiO<sub>2</sub>, the heterojunction of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd ternary nanocomposite shows the larger visible light absorption region, which should improve photocatalytic activity as a stronger absorption intensity tends to improve the performance of photogenerated carriers. Tauc plots for TiO<sub>2</sub>, TiO<sub>2</sub>/Pd,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> samples are shown in Fig. 3b-c. Tauc plots of α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd in Fig. 3d confirms the reduction of the band gap energy in the ternary heterojunction α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd nanocomposite.

The band gaps and corresponding valence band potential  $(E_{VB})$  were determined from the XPS spectra of TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> following a procedure reported by Bhoi et al. [63]. The  $E_{VB}$  is 2.78 eV and 1.39 eV for TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, respectively. The conduction band potential  $(E_{CB})$  value of TiO<sub>2</sub> is more positive than that of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (-0.27 eV and -0.63 eV relative to the Fermi level) [63, 64]. Therefore, the energy level potentials of TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> in photocatalyst could be combined to form a heterojunction of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> for photogenerated charge-separation and transfer [65].

The flat band potentials ( $V_{fb}$ ) of TiO<sub>2</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd as determined by Mott-Schottky plots (Fig S2) were 0.73, -0.31, -0.33 and -0.37 V (vs. Ag/ AgCl), respectively. The slopes of all photocatalyzed samples (except TiO<sub>2</sub>) are negative, which corresponding to the p-type of semiconductor [8]. The V<sub>fb</sub> (vs. Ag/AgCl) obtained from the reference electrode was conversed to reversible hydrogen electrodes (RHE) and normal potential (NHE) unit



 $\textbf{Fig. 3} \quad \textbf{a} \text{ UV-Vis absorption spectra; and Tauc plots of } \textbf{b} \text{ TiO}_2 \text{ and } \text{TiO}_2/\text{Pd}, \textbf{c} \\ \alpha \text{-} \text{Fe}_2 O_3, \\ \alpha \text{-} \text{Fe}_2 O_3/\text{TiO}_2, \\ \textbf{and } \textbf{d} \\ \alpha \text{-} \text{Fe}_2 O_3/\text{TiO}_2. \\ \textbf{Pd} \\ \textbf{c} \\ \alpha \text{-} \text{Fe}_2 O_3, \\ \alpha \text{-} \text{Fe}_2 O_3/\text{TiO}_2. \\ \textbf{c} \\ \alpha \text{-} \text$ 

 Table 2
 The flat band potential, conduction band, and valence band values as obtained from Mott-Schottky plots

Samples	Flat band potential (V <sub>fb</sub> ) (vs. Ag/AgCl)	Conduction band (E <sub>CB</sub> ) (vs. NHE)	Valence band (E <sub>VB</sub> ) (vs. NHE)
TiO <sub>2</sub>	0.73	0.92	4.09
α-Fe <sub>2</sub> O <sub>3</sub>	-0.31	-0.11	1.91
α-Fe <sub>2</sub> O <sub>3</sub> /TiO <sub>2</sub>	-0.33	-0.13	2.05
α-Fe <sub>2</sub> O <sub>3</sub> /TiO <sub>2</sub> - Pd	-0.37	-0.17	1.83

following the procedure advised in a previous study [16]. Table 2 presents the potential of conduction band (E<sub>CB</sub>) of TiO<sub>2</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd, which corresponds to 0.92, -0.11, -0.13, and -0.17 (vs. NHE), respectively. The potential of valence band (E<sub>VB</sub>) of TiO<sub>2</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> and Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd was determined to be 4.09, 1.91, 2.05, and 1.83 (vs. NHE), respectively. The smaller E<sub>CB</sub> and E<sub>VB</sub> values obtained in  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd sample indicates the superior capability to enhance charge



Fig.4 Nitrogen adsorption–desorption isotherms of TiO\_2,  $\alpha\text{-}Fe_2O_3/\text{TiO}_2$  and  $\alpha\text{-}Fe_2O_3/\text{TiO}_2\text{-}Pd$ 

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carrier separation and the decrease in driving force for a specific photocatalytic reaction [8, 16].

The nitrogen adsorption-deposition isotherms are shown in Fig. 4. The textural properties including BET specific surface area, average pore diameter, and average pore volume for the photocatalyst samples are summarized in Table 3. The isotherm obtained from the pure TiO<sub>2</sub> fit well with the type V isotherm curve, while the isotherms of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd exhibited a type IV isotherm. These

 $\ensuremath{\text{Table 3}}$  BET surface area, average pore diameter, and average pore volume of samples

Sample	BET surface area (BET, m <sup>2</sup> /g)	Average pore diameter (nm)	Average pore volume (cm <sup>3</sup> /g)
TiO <sub>2</sub>	15.68	8.22	0.032
TiO <sub>2</sub> /Pd	13.25	9.41	0.031
α-Fe <sub>2</sub> O <sub>3</sub> /TiO <sub>2</sub>	21.05	12.62	0.066
$\alpha$ -Fe <sub>2</sub> O <sub>3</sub> /TiO <sub>2</sub> -Pd	25.51	14.20	0.090

results are similar to the results obtain by Khasawneh et al. [51]. The hysteresis loops are H3 type according to IUPAC classification at the high relative pressure for  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd [66]. This result indicates that all types of catalyst contain mesoporous structure with the narrow pore size between 8 and 14 nm. [66, 67]. The BET specific surface area (SSA) was 15.68, 21.05 and 25.51 m<sup>2</sup>/g for TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>, and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd, respectively (Table 3). The incorporation of Pd ternary with TiO<sub>2</sub> has significantly increased SSA, crystallinity, and surface activity on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. Meanwhile, 1%Pd incorporation resulted in 13% and 27% increase in the average pore size and the average pore volume, respectively. Therefore, the highest BET specific area and pore diameter observed in the ternary of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd indicates an improve in photocatalyst properties.

The recombination and separation efficiency of charge carriers was investigated from PL spectra as shown in Fig. 5a. The strong PL intensity was observed for pure  $TiO_2$ , indicating that the electrons and holes recombined rapidly [44, 68]. The emission intensity of photocatalyst



 $\label{eq:Fig.5} a PL spectra of TiO_2, \alpha-Fe_2O_3/TiO_2 and \alpha-Fe_2O_3/TiO_2-Pd, \mbox{b} Photocurrent response curve, and \mbox{c} EIS Nyquist plots of TiO_2, \alpha-Fe_2O_3/TiO_2 and \alpha-Fe_2O_3/TiO_2-Pd under visible light irradiation and (Compared to the second to the s$ 

heterojunction  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> nanocomposite was lower than that of pure TiO2, indicating that the recombination rate of electron-hole pairs was reduced for photocatalyst nanocomposite sample. A lower PL intensity indicates lower recombination rate and higher photocatalyst activity [64, 69]. After loading Pd nanoparticles on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/ TiO<sub>2</sub> nanocomposite, the emission intensity of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/ TiO<sub>2</sub>-Pd was lower than that of α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> photocatalyst, which indicates that the Pd effectively inhibited the  $e^{-}h^{+}$  recombination [70]. Therefore, the loading of Pd nanoparticles can improve charge separation efficiency in photocatalytic applications. To confirm the separation and migration, photogenerated charge carriers in pure TiO2,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd photocatalyst materials were measured by photocurrent response as shown Fig. 5b. The photocurrent of the pure TiO2 was found to be the lowest, indicating the rapid recombination of photogenerated charge carrier. The photocurrent response of the ternary heteroiunction  $\alpha$ -Fe<sub>2</sub>O<sub>2</sub>/TiO<sub>2</sub>-Pd nanocomposite was much greater than that of pure TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> ( $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/  $TiO_2$ -Pd >  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> > TiO<sub>2</sub>), which is an evidence of a more efficient separation of photogenerated e<sup>-</sup>-h<sup>+</sup> pair [64, 67]. The EIS Nyquist of TiO<sub>2</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/ TiO<sub>2</sub>-Pd were 17.2 k $\Omega$ , 15.7 k $\Omega$  and 10.3 k $\Omega$ , respectively (Fig. 5c). The EIS Nyquist plot of TiO<sub>2</sub> exhibited a higher semicircle curve, suggesting the highest resistance of charge transfer compared to those of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/ TiO<sub>2</sub>-Pd. The α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd curve had small radius circle, for which indicates a lower resistance for charge transfer and a lower  $e^--h^+$  recombination rate [45, 57]. It can be concluded that Pd nanoparticle deposited on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> could provide a more efficient interfacial charge transfer according to the mesostructured, high SSA, and at the same time also reduce the e<sup>-</sup>-h<sup>+</sup>recombination.

The morphology of TiO<sub>2</sub>, Fe<sub>2</sub>O<sub>3</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd nanocomposites was characterized by FE-SEM and TEM. In Fig. 6a, the morphology of TiO2 shows an average diameter in the range of 50-100 nm and an average length of 5-7 µm, which agrees well with number published in prior reports [38, 71]. The morphology of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> is shown in Fig. 6b, for which illustrate the average diameter of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> of 500 nm. According to a previous study [50], for  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/ TiO<sub>2</sub> nanocomposite, α-Fe<sub>2</sub>O<sub>3</sub> was deposited densely and combined tightly onto the TiO<sub>2</sub> surfaces to form heterojunctions between these two materials. In this study, we found that the Pd composite did not cause any significant change in the structure of the heterojunctions of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> nanocomposite as observed in Fig. 6c and d. A typical TEM image of heterojunction α-Fe2O3/TiO2-Pd ternary nanocomposites (Fig. 6e) and the existence of the lattice spacings of 0.35 and 0.25 nm (Fig. 6f-h) can be attributed to (101) and (110) crystal planes of anatase TiO2 and α-Fe2O3, respectively [49]. Meanwhile, Pd demonstrated lattice spacing of

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Fig. 6 FE-SEM images of a TiO<sub>2</sub>, b  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, c, d  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd, and e-h TEM images of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd

0.22 nm that corresponds to the (111) plane of the metal itself [61].

The EDS results of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd photocatalytic nanocomposites are presented in Fig. 7a, b, which confirm the presence of Fe and Pd elements on the surfaces of TiO<sub>2</sub>. The elemental composition of Ti, O, C, Fe, and Pd were found to be 42.4, 36.6, 10.9, 9.5, and 0.7 wt% respectively. The elemental EDS mapping images of Ti, Fe, and Pd in Fig. 7c-e clearly demonstrate that TiO<sub>2</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, and Pd were uniformly distributed on the photocatalyst surfaces.

The chemical states of elements in the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd ternary heterojunction photocatalysts were measured by using high-resolution XPS. In Fig. 8a, the survey scan illustrates the presence of Ti, O, Fe, and Pd elements. The high-resolution Ti scan in Fig. 8b indicates two binding peaks at 464.2 and 458.5 eV, which correspond to Ti 2p<sub>1/2</sub> and Ti 2p<sub>3/2</sub>, respectively. The split between Ti 2p<sub>1/2</sub> and Ti 2p<sub>3/2</sub> (5.7 eV) indicated the normal state of Ti<sup>4+</sup> [37, 72]. The Ti 2p<sub>1/2</sub> peak at 458.5 eV was caused by the

S-Scheme  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> Photocatalyst with Pd Cocatalyst for Enhanced Photocatalytic..



formation of Fe–O–Ti bonds in the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> [73]. In Fig. 8c, the binding energies of Fe at 711.3 and 724.8 eV correspond to Fe 2p<sub>3/2</sub> and Fe 2p<sub>1/2</sub> of Fe<sub>2</sub>O<sub>3</sub>, respectively. A shakeup satellite peak at 719.1 eV occurred due to the Fe<sup>3+</sup> in  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> [19]. In Fig. 8d, the Pd 3d spectrum shows two peaks with the binding energies of 338.1 and 343.5 eV, which are assigned to Pd 3d<sub>5/2</sub> and Pd 3d<sub>3/2</sub> of metallic Pd [74], respectively. Hence, the Pd element was successfully deposited onto the heterojunction of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> as a ternary. The O 1s spectrum displayed component lattice cantered at 529.1 eV, and two binding peaks at 531.3 and 532.2 eV, which attributed to oxygen species (OH) (Fig S1) [50] bonded to Ti–O (titanium) and Fe–O (iron) [38, 50]. All of the above results indicate that the ternary heterojunctions in  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd nanocomposite were synthesized successfully.

### 3.2 H<sub>2</sub> Production

The photocatalytic H<sub>2</sub> production rates from water and a sacrificial agent over different TiO<sub>2</sub>-based photocatalysts are presented in Fig. 9a. The H<sub>2</sub> production rate was at 40.12  $\mu$ mol.h<sup>-1</sup> g<sup>-1</sup> for anatase TiO<sub>2</sub>. This low production rate indicates poor photocatalytic activity of pure TiO<sub>2</sub>. The

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Fig. 8 XPS survey spectrum of  $\mathbf{a} \propto -Fe_2O_3/TiO_2$ -Pd and high-resolution spectra of  $\mathbf{b}$  Ti 2p,  $\mathbf{c}$  Fe 2p, and  $\mathbf{d}$  Pd 3d

 $\rm H_2$  production rate of 1400.91  $\mu mol.h^{-1}~g^{-1}$  was achieved for TiO\_2/Pd with UV light. After loading  $\alpha$ -Fe\_2O\_3 onto the TiO\_2 surface, the production rate of 740.60  $\mu mol.h^{-1}~g^{-1}$  was obtained. In comparison, ternary heterojunction  $\alpha$ -Fe\_2O\_3/TiO\_2-Pd samples showed a significantly higher H\_2 production rate at 3490.54  $\mu mol.h^{-1}~g^{-1}$ , corresponds to an increase of 78.65% compared to  $\alpha$ -Fe\_2O\_3/TiO\_2. Hence, the highest H\_2 production rate was observed with the ternary heterojunction  $\alpha$ -Fe\_2O\_3/TiO\_2-Pd nanocomposite.

The effects of irradiation time on H<sub>2</sub> production using different samples are presented in Fig. 9b. The pure TiO<sub>2</sub> anatase corresponded to a low H<sub>2</sub> production rate at due to its rapid recombination of photogenerated  $e^-$ -h<sup>+</sup> pairs [75]. The  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> also exhibited a lower photocatalyst activity in the beginning because of the recombination of photogenerated charge carriers, but the H<sub>2</sub> production gradually increased to 11,971 µmol g<sup>-1</sup> after 6 h of irradiation. For  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd, the H<sub>2</sub> production increased rapidly over irradiation time and reached 27,072 µmol g<sup>-1</sup> after 6 h.

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These results of Pd acting as a co-catalyst enhancing photocatalysis activity were based on electron transfer charge carrier and separation functions.

After 4-h cycle of  $\mathrm{H}_{2}$  production, the photocatalyst was cleaned and dried to be reused again. The stability of ternary α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd photocatalyst in H<sub>2</sub> production was evaluated and the results are shown in Fig. 9c. After 5 cycles over the 20-h period,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd photocatalyst still exhibited a good photocatalytic stability in H<sub>2</sub> production. Thus, the presence of ternary Pd improved the stability of H<sub>2</sub> production. Liu et al. [50] also reported that the Fe2O3-TiO2-Pt nanocomposite had excellent stability in H2 production. However, the  $H_2$  production rate for  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/ TiO<sub>2</sub>-Pd photocatalyst slightly decreased with reaction cycles. 10% reduction after five cycles was observed. This finding is in accordance with a previous study [50] reporting that the photocatalytic rate was slightly reduced during the long-term reaction, possibly by the formation of Pd oxide [70].



Fig.9 a Hydrogen evolution from different photocatalysts, b hydrogen production over a long irradiation time, c stability of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd after recycling, and d XRD assessment after recycling

The XRD patterns of the virgin  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd photocatalyst and the one after 5 consecutive cycles of photocatalytic reactions are shown in Fig. 9d. There is no drastic change observed between the characteristic peaks of the virgin and used samples, which suggesting that these photocatalyst nanocomposites have good long-term stability and reusability [44]. We concluded that the presence of Pd ternary deposition over the heterojunction of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> enhanced the photocatalytic activity as well as the stability of photocatalyst.

Table 4 compares H<sub>2</sub> production rate from the results of several studies under similar experimental conditions but with different noble metals used as co-catalysts. The H<sub>2</sub> production rate from Pt ternary loaded onto Tio<sub>2</sub>–Fe<sub>2</sub>O<sub>3</sub> nanocomposite under visible light irradiation and MeOH as sacrificial agent was at 90 µmol h<sup>-1</sup> [50], while Au [76] or Pd [77] loaded onto TiO<sub>2</sub>/graphene nanocomposite corresponded to H<sub>2</sub> production rate of 296 and 280.8 µmol h<sup>-1</sup>, respectively. The hydrogen production rate measured in this study is higher than those in previous studies [76, 77] as the loaded ternary Pd on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> heterojunction induced a higher photocatalytic activity. Moreover, the Pd loading

could reduce the recombination of  $e^-h^+$ , producing a higher charge transfer rate coefficient [37]. Therefore, it can be concluded that using Pd as a co-catalyst effectively enhances  $H_2$  production.

In general, when heterojunction photocatalysts are developed from more than two semiconductors, they do not only allow a broad range of light absorption but also enhance charge separation by separating the electrons and holes to different semiconductors. The developed heterojunction can be categorized into type-II heterojunction or stepscheme (S-scheme) heterojunction. In this study, the possible mechanism is that  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and TiO<sub>2</sub> form S-scheme heterojunction. The S-scheme heterojunction differs from a type-II heterojunction, as the charge-transfer mechanism of the S-scheme heterojunction comprise a reduction photocatalyst and an oxidation photocatalyst with staggered band structure [78].

As the work function of heterojunction photocatalyst was investigated, the interfacial charge transfers between  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and TiO<sub>2</sub> of before contact, after contact, and light irradiation are shown in Fig. 10. Before contact, the work function of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and TiO<sub>2</sub> was previously reported to be

Table 4 $H_2$ production ratewith different noble metals asco-catalysts on heterojunction	Heterojunction	7
photocatalysts	α-Fe <sub>2</sub> O <sub>3</sub> /TiO <sub>2</sub>	F
	CdS/g-C <sub>3</sub> N <sub>4</sub>	F

Heterojunction	Ternary	Sacrificial agent	Light Source	$H_2$ production (µmol h <sup>-1</sup> )	Refs
α-Fe <sub>2</sub> O <sub>3</sub> /TiO <sub>2</sub>	Pd	Methanol	250 W, LED,	349.5	This work
			$\lambda \ge 420 \text{ nm}$		
CdS/g-C <sub>3</sub> N <sub>4</sub>	Pd	Sodium sulfite	300 W, Xe	293.0	[45]
			$\lambda \ge 400 \text{ nm}$		
TiO2-Fe2O3	Pt	Methanol	300 W, Xe,	90.0	[ <mark>50</mark> ]
		$\lambda \ge 400 \text{ nm}$			
TiO <sub>2</sub> /GO	Au	Methanol	12 W, LED,	296.0	[ <b>76</b> ]
			$\lambda \ge 420 \text{ nm}$		
TiO <sub>2</sub> /GO	Pd	Methanol	288 W, Hg	280.8	[77]
-			$\lambda \ge 400 \text{ nm}$		
TiO <sub>2</sub> /g-C <sub>3</sub> N <sub>4</sub>	Pt	Triethanolamine	300 W, Xe	40.0	[81]
			$\lambda \ge 420 \text{ nm}$		
TiO <sub>2</sub> -ZnO	Pt	Methanol	300 W, Hg	215.0	[83]
			$\lambda \ge 420 \text{ nm}$		
TiO <sub>2</sub> /V <sub>2</sub> O <sub>5</sub>	Pt	Methanol	125 W, Hg,	296.6	[84]
2 2 9			$\lambda \ge 400 \text{ nm}$		

GO graphene oxide, Xe xenon lamp, Hg mercury lamp,  $V_2O_5$  vanadium oxide, g- $C_3N_4$  graphitic carbon nitride, ZnO zinc oxide and  $Na_2S$  sodium sulphide



Fig.10 S-scheme  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and TiO<sub>2</sub> pathway of charge carrier: **a** the work function before contact, **b** after contact and **c** light irradiation of S-scheme heterojunction

5.88 eV [79] and 7.18 eV [80], respectively (Fig. 10a). After contact (Fig. 10b), the difference of work function indicated the electron transfer from  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> to TiO<sub>2</sub> as Fermi energy of TiO<sub>2</sub> was lower than that of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> [79, 80]. TiO<sub>2</sub> has a greater work function, which indicates that charge transfer occurred at the interface from  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> to TiO<sub>2</sub>, relative to the equilibrium of the Fermi energies [40]. The charge transfer led to an increase of electrons density at TiO<sub>2</sub> and a decrease of electrons density at  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> at the interface. This result indicates a built-in electric field between  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and TiO<sub>2</sub> at the interface. Under light irradiation (Fig. 10c),

electrons are excited from CB of TiO<sub>2</sub>, then the electrons were transferred to VB of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and recombined with the hole in VB of Fe<sub>2</sub>O<sub>3</sub>. Simultaneously, the recombination of electrons and hole in CB of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and VB of TiO<sub>2</sub> can be prevented, for which enhanced the photogenerated charge carrier [58].

The mechanism of heterojunction electron transfer in ternary of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd under visible light irradiation for H<sub>2</sub> generation is shown in Fig. 11. When visible light is irradiated on the heterojunction of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd nanocomposite, the photogenerated electrons in the CB of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> are transferred

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S-Scheme  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> Photocatalyst with Pd Cocatalyst for Enhanced Photocatalytic..

Fig. 11 Mechanism of photocatalysis over α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd under visible light: a type-II heterojunction and b S-scheme heterojunction

to the CB of TiO<sub>2</sub>, while the holes show a reverse transfer route [45, 78]. Furthermore, these electrons migrate to Pd on the crystallite surface in the CB of TiO2 and accumulate there, forming a Schottky barrier between the Pd nanoparticle as ternary on the surface of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> [44, 81]. The photogenerated holes in the VB reacted with water to create pairs of the proton (H<sup>+</sup>) and hydroxyl radical (·OH) [33]. The generated H<sup>+</sup> will recombine with free electrons from Pd to form H atom and subsequently the H2 molecule. By allowing the ·OH to react with the sacrificial agent methanol, the timescale of separation for the photogenerated electrons and holes is prolonged. TiO<sub>2</sub> has a larger work function (7.18 eV) with an oxidationtype, while  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (5.88 eV) has a smaller work function with reduction-type of photocatalysts. The mechanism of ternary of α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd as explained by the S-scheme heterojunction is that the electron in the VB of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and TiO<sub>2</sub> are transferred to CB of each photocatalyst. Meanwhile, the electrons in the CB of TiO<sub>2</sub> are easily transferred to the VB of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, because the driving force of the electric field at in the interface [80, 82]. Thus, the photogenerated electrons in CB of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> can migrate to Pd, which increase the redox ability for H<sub>2</sub> production, the photogenerated h<sup>+</sup> in the VB of TiO2 involves H2O oxidation and increase the separation of photogenerated e<sup>-</sup>-h<sup>+</sup> pairs based on the results discussed earlier. Therefore, the results indicate that the ternary heterojunction of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd was a typical S-Scheme system as shown in Fig. 11b.

### 4 Conclusions

In summary, the new S-scheme  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> heterojunctions with ternary Pd nanoparticles were successfully synthesized in this work to use in H<sub>2</sub> production through water splitting.

- The heterojunction α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd ternary nanocomposites achieved absorption in the range of visible light, high-efficiency in the separation of e<sup>-</sup>-h<sup>+</sup> pairs, and stable performance in the multi-cycle reaction for H<sub>2</sub> production. The Pd ternary reduced the e<sup>-</sup>h<sup>+</sup> recombination rates, which leads to higher charge transfer and H<sub>2</sub> production rates.
- The ternary Pd nanoparticle loading on heterojunction α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> exhibited great performance, with H<sub>2</sub> production rate of 3490.45 µmol h<sup>-1</sup> and excellent stability in multi-cycle H<sub>2</sub> production. Therefore, the experimental results and theoretical analyses confirm the high potential for the applicability of the new photocatalysts for hydrogen production through photocatalytic water splitting.

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S-Scheme  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> Photocatalyst with Pd Cocatalyst for Enhanced Photocatalytic...

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### **CHAPTER 3**

This chapter presents the modification of heterojunction photocatalyst of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> nanocomposite by Ar-plasma treatment for hydrogen evolution. The  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> has obtained a characterization of the crystal structure, optical properties, bandgap energy, surface morphology, elemental composition and charge carrying capacity of the photocatalyst nanocomposite was investigated. Finally, the performance of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> for the H<sub>2</sub> production and photocatalyst stability was investigated. This article has been published in the title of "Modification of surface  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> photocatalyst nanocomposite with enhanced photocatalytic activity by Ar gas plasma treatment for hydrogen evolution" in the Journal of Environmental Chemical Engineering.

## PAPER II

Modification of surface  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> photocatalyst nanocomposite with enhanced photocatalytic activity by Ar gas plasma treatment for hydrogen evolution.

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Modification of surface  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> photocatalyst nanocomposite with enhanced photocatalytic activity by Ar gas plasma treatment for hydrogen evolution

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### ABSTRACT

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Hydrogen evolution Ar-plasma treatment

In this work, hematite (α-Fe<sub>2</sub>O<sub>3</sub>) deposited on anatase TiO<sub>2</sub> was modified by plasma treatment and studied. a-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> was successfully synthesized using sol-gel precipitation methods and the surface was modified by argon (Ar) gas plasma treatment to increase photocatalyst nanocomposite activity. The effect of Ar-plasma treatment times from 5 to 20 min at DC glow discharge powers of 50 watts was investigated. The crystal structures, surface morphology, optical properties were characterization by using X-ray Diffractometry (XRD), scanning electron microscopy (SEM), transmission electron microscopy (TEM), atomic force microscopy (AFM), UV–vis spectrometry (UV–Vis), X-ray photoelectron spectroscopy (XPS), photoluminescence spectroscopy (PL) and electrochemical impedance spectroscopy (EIS). After plasma treatment, the crystalline phase of  $TiO_2$  form changed from pure anatase to a mixture of anatase and rutile phases. The particle size was reduced from 25 to 20 changed from pure anatuse to a mature of anatuse and rutue phases. The particle size was reduced from 25 to 20 nm. The band gap energy decreased from approximately 2.18 eV to 1.84 eV and exhibited a wider absorption edge under visible light irradiation (400–700 nm). Ar-plasma modification of the surface showed the attribution of oxygen vacancies and  $TI^{3+}$ . The plasma treatment improved the heterojunction photocatalyst to enhance the photocatalytic activity under visible light for H<sub>2</sub> evolution. The best photocatalytic activity for H<sub>2</sub> evolution rate was obtained at 1.25 mmol h<sup>-1</sup> for  $\alpha$ -fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10 photocatalyst compared to  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-untreated (0.13 mmol h<sup>-1</sup>). The plasma treatment on photocatalyst materials showed long-term stability and excellent separation of the photogenerated charge carriers

#### 1. Introduction

Currently, energy sources from fossil fuels are limited and cause environmental pollution, which is a problem that can be solved by renewable energy [1-5]. Renewable energy sources from both solar and wind energy are considered a promising technology for cheap and un-limited energy from natural resources [6-8]. Hydrogen (H<sub>2</sub>) is an ideal solution to provide energy in the future because of its high efficiency and cleanliness [6,9]. There are several methods for  $H_2$  production from solar light, including photoelectrochemical [6], photocatalysis [5], photoelectrolysis [6] and photobiological [10]. Photocatalytic splitting of water was proposed to produce  $H_2$  under solar light irradiation because it is a sustainable energy source that is environmentally friendly

Photocatalytic water splitting for H<sub>2</sub> evolution was reported by using titanium dioxide (TiO<sub>2</sub>) as a semiconductor material.TiO<sub>2</sub> nanoparticle semiconductor materials have been widely studied in photocatalysis because of high stability, specific morphology and low toxicity [12,13]. However, anatase TiO2 phase has a wide band gap (3.2 eV) of photocatalytic activity and is active only under UV light irradiation with fast recombination of electron and hole pair (e', h<sup>+</sup>) [3,14,15]. In general, of the three crystalline phases of  $\mathrm{TiO}_2$  (anatase, rutile and brookite), anatase phase is the most used as a photocatalyst due to less recombination of the electron-hole pair when compared with other phases [16–18]. The crystalline structure of mixed phase anatase  $TiO_2$  (3.2 eV) and rutile  $TiO_2$  (3.02 eV) [19] and the commercial Degussa P25 commonly show higher performance of photocatalytic activity [20]. It has reduced band gap energy, increased absorption under visible light

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### and enhanced photoactivity.

Photocatalytic activity of TiO<sub>2</sub>-based photocatalysts can gain enhanced absorbance of visible light and increased separation of the electron-hole pair with metal deposition as a light-scattering material, such as TiO<sub>2</sub>/WO<sub>3</sub> [21], TiO<sub>2</sub>/ZnO [22], TiO<sub>2</sub>/SnO<sub>2</sub> [23], Cu<sub>2</sub>O/TiO<sub>2</sub> [24] and TiO<sub>2</sub>/Fe<sub>2</sub>O<sub>3</sub> [25], etc. For illustration, Lee et al. [23] found that the composite TiO<sub>2</sub>/SnO<sub>2</sub> exhibited high visible light harvesting and could be applied as a photocatalyst for H<sub>2</sub> evolution. Yan et al. [24], studied the composite TiO<sub>2</sub>/Cu<sub>2</sub>O and found a high efficiency in charge transfer of electron hole rate and a high H<sub>2</sub> evolution rate (16.25 mmol  $h^{-1} g^{-1}$ ).

Heterojunction nanocomposites for photocatalysis are desired for their higher efficacy, reuse ability, environmental friendliness, high stability and low cost. Increased photocatalyst efficiency is currently in focus for the application of water splitting and wastewater treatment. One of the noble metals that can be deposited over  $TiO_2$  is hematite ( $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>), a metal oxide with a narrow bandgap of about 2.0–2.2 eV [26]. It seems to be a suitable semiconductor photocatalyst for H2 evolution, with efficiency harvesting in the solar spectrum (43%), excellent stability, non-toxic and is cost-effective [27-3 30]. Several studies (Madhumitha et al. [27], Fu et al. [31], Mohamed et al. [32], Lu et al. [33], Zhu et al. [34]) have demonstreted that  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> deposition on TiO<sub>2</sub> nanoparticles have effective charge separation, enhanced visible light absorption and enhanced photocatalytic activity for H<sub>2</sub> evolution. The methods that have been reported for the fabrication of photocatalytic  $\alpha\text{-}Fe_2O_3/\text{Ti}O_2$  nanocomposite heterojunction include sol-gel [35], hydrothermal and impregnation methods [36,37]. To improve the photocatalytic activity of TiO2-based photocatalysts, various fabrication methods have been tried, but remain a challenge. The drawback of  $\alpha\text{-}\text{Fe}_2\text{O}_3$  is the limited performance and high recombination rate of the photogenerated electrons and hole [38,39]

The plasma process was reported to modify the surface properties of photocatalysts, where the presence of highly stable Ti3+ oxygen vacancy could improve the performance of photogenerated charge carriers [40,43,44]. The advantage of plasma treatment is low energy consumption and not requiring any toxic chemicals [45]. In previous studies, Kong et al. [46] studied Ar plasma treatment could cause oxygen vacancy defects over a TiO2/GO heterojunction that improved photocatalyst activity and separation of the photogenerated electron-hole pair for  $H_2$  evolution. Xing et al. [47] found that Ar plasma modification of surface Pt/g-C3N4 nanosheets can improve charge transfer of photogenerated performance for enhanced H<sub>2</sub> evolution in photocatalytic activity. So Ar-plasma can be used to produce oxygen vacancy defects in a metal oxide, one of the most stable methods of creating Ti<sup>3+</sup> in TiO<sub>2</sub>. For example, Sun et al. [48] modified the surface of TiO2/CoFe2O4 using Ar-plasma and showed that changing uniform TiO<sub>2</sub> to anatase and rutile phase produced oxygen vacancy defects on the surface which improved the photocatalytic activity. Kawakami et al. [49] have modified anatase  $\rm TiO_2$  by Ar gas, they found the mix of anatase and rutile phase increased with plasma treatment time. Plasma methods do not require high temperatures to modify the structure of TiO<sub>2</sub> anatase to mixed anatase and rutile phase, which is a benefit compared to annealing methods that typically use calcination at temperatures of 450-900 °C for at least 2 h [50]. Therefore, Ar plasma treatment can be used to improved properties of TiO<sub>2</sub> crystalline structure, alter the phase to mixed anatase and rutile phase and enhance photocatalytic activity with less recombination of electron-hole pairs.

The photocatalytic activity of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> nanocomposite for H<sub>2</sub> evolution by Ar-plasma treatment have not been previously reported. In this work,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> was loaded onto TiO<sub>2</sub> by using the sol-gel precipitation methods and the surface of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> was modified by Artreatment at several exposure times (5–20 min) to improve the photocatalytic activity. The crystal structure, optical properties, bandgap energy, surface morphology, elemental composition and charge carrying capacity of the photocatalyst nanocomposite was investigated. Finally, the performance of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> for the H<sub>2</sub> evolution and photocatalyst stability was investigated.

#### 2. Material and methods

### 2.1. Preparation of TiO<sub>2</sub>

TiO<sub>2</sub> nanoparticle was synthesized according to previous studies by Peng et al. [51]. Titanium (IV) butoxide (Ti (OC(CH<sub>3</sub>)<sub>3</sub>)<sub>4</sub>, Sigma-Aldrich, USA), 3 mL, was mixed with ethylene glycol (Loba Chemie, India), 30 mL, in a bottle flask. The solution of pure TiO<sub>2</sub> was treated at 180 °C with magnetic stirring for 2 hrs. The white precipitate of TiO<sub>2</sub> was centrifuged at 5000 rpm (Force 7, Denver Instrument, USA) and washed with ethanol (EtOH, Ar 99.9%, RCI Labscan, Thailand) and deionized water several time. Finally, the sample was calcined at 450 °C for 2 hrs to a stable pure TiO<sub>2</sub> dry powder.

#### 2.2. Preparation of $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> on Ti substrate

The synthesis of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> on TiO<sub>2</sub> nanocomposite photocatalyst is show in Fig. 1. Firstly, pure TiO<sub>2</sub> powder (1.5 g) from Section 2.1 and 30 mL of iron (III) nitrate nonahydrate Fe (NO<sub>3</sub>)<sub>3</sub>·9H<sub>2</sub>O, (Ajax Finecem, Australia) 50 wt% was mixed in ethanol under magnetic stirring for 30 min and evaporated at 50 °C.

To deposit  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> nanocomposite on Ti substrate,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> (1.5 g) powder was dispersed in 50 mL of DI water. After that, 0.06 g Polyethylene glycol (PEG, Mw 6000, Ajax Finecem, Australia) was dropped into the above solution and mixed under stirring for 30 min. The coating was formed on the Ti (3  $\times$  5 cm) substrate by the dip-coating method. Finally, Ti substrates were immersed in the aqueous solution for 10 min and then dried in a vacuum oven at 120 °C for 30 min

### 2.3. Plasma procedure

The plasma treatment process used Argon (Ar) (purity 99.9%; flow rate, 50 sccm) gas flow in a plasma reactor [52]. A DC power supply of 50 W was used to generate plasma discharges. Plasma treatment exposure times were 5, 10, 15 and 20 min to produce photocatalytic nanocomposite materials, referred to as  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-5,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-15 and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-20, respectively. A schematic illustration of the plasma procedure is shown in Fig. 2.

#### 2.4. Characterization of materials

The surface morphology was measured by a scanning electron microscope (SEM, Quanta 400, Thermo Fisher Scientific, USA). Morphology was determined by Transmission electron microscopy (TEM, Tecnai G2 20 S-TWIN 200 KV, FEI, USA). The chemical component surface was analyzed by X-ray photoelectron spectroscopy (XPS, AXIS Ultra DLD, Krotos Analytical, UK). The crystal structure was determined using an X-ray Diffractometer (XRD, X' Pert MPD, PHILIPS, Netherlands) with a Cu K $\alpha$  source ( $\lambda=0.1540$ Å) scanning at the 20 range from 20° to 80°. The band gap was calculated by a UV–vis spectrometer (UV-Shimadzu, Japan). The surface roughness was measured using an Atomic Force Microscope (AFM, Nanosurf easy Scan2, Switzerland). The electron-hole pair recombination rate of the photocatalyst was measured by a photoluminescence spectrometer (PL; Avantes AvaSpec-2048, Netherlands) at an excitation wavelength 590 nm.

### 2.5. Electrochemical impedance spectroscopy (EIS)

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The electrochemical impedance spectroscopy (EIS) was performed by using a Potentiostat/Galvanostats (PGSTAT302N, Metrohm, Autolab, Netherland) workstation with three electrode cells. The fluorine-doped tin oxide (FTO) electrode performed deposition with the photocatalyst

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Fig. 1. The synthesis of photocatalyst of the  $\alpha\mbox{-}Fe_2O_3$  on  $TiO_2$  nanocomposite



Fig. 2. Schematic illustration of the plasma-treated on α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> photocatalytic nanocomposite.

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samples, platinum wire and a Ag/AgCl electrode was used as the counter electrode and reference electrode, respectively [53]. EIS was performed at voltage amplitude of 10 mV with a frequency range of 0.01–1,000, 000 Hz with 0.5 M Na<sub>2</sub>SO<sub>4</sub> aqueous solution as the electrolyte [43]. Fabrication of photocatalyst samples for the working electrodes were prepared as follows: 0.05 g of photocatalyst sample was mixed with 1 mL ethanol for 20 min and deposited on 1 × 2 cm<sup>2</sup> FTO glass electrodes and dried at 80 °C for 6 h. The light source was a 250 W mercury lamp (12750 lumens, Philips, NPL-N).

### 2.6. H<sub>2</sub> evolution

The photocatalytic activity reactor was set up in a plexiglass cell with a volume 500 mL. A mercury lamp 250 W (12750 lumens, Philips, NPL-N) was used as a visible light source positioned 15  $\rm cm^2$  away from the plexiglass on the opposite side to the photocatalytic reactor. The photocatalyst samples of TiO<sub>2</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> and plasma-treated  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/ TiO<sub>2</sub> (α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-5, α-Fe<sub>2</sub>O<sub>3</sub>TiO<sub>2</sub>-10, α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-15 and α-Fe<sub>2</sub>O<sub>3</sub>/ TiO\_2-20) were put into the photocatalyst reactor with a sacrificial agent solution in methanol: water (30: 70 v/v). Before photocatalyst testing in the photocatalytic reactor, argon (Ar purity 99.99%) gas was flowed into the reactor for 30 min to completely remove air. The amount of H2 concentration was analyzed by gas chromatography (GC; Hewlett Packard, HP 6890, USA) equipped with a thermal conductivity detector (TCD). For the analysis, gas was injected through 5 Å molecular sieve stainless steel columns and helium (He) gas was used as a carrier gas. The concentration of hydrogen was calculated using standard curve calibration. The H<sub>2</sub> evolution rate was calculated following Heider and Kang [54]. The recycling stability of the photocatalyst reaction was tested for each sample and repeated five times for each cycle [55]. Ferric ion and titanium leaching of the photocatalytic reaction after used for each recycle time was determined by an inductively coupled plasma optical emission spectrometry (ICP-OES, Avio 500, Perkin Elmer, USA)

### [56].

### 3. Results and discussion

3.1. a-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> photocatalyst nanocomposite characteristic

XRD analysis of the crystalline characteristics of the plasma modified  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> deposited on anatase TiO<sub>2</sub> nanoparticle shows the anatase TiO<sub>2</sub> as representative diffraction peaks at  $2\theta=25.3^\circ, 37.8^\circ, 48.14^\circ, 53.9^\circ$  and 54.1° (Fig. 3), which is attributed to the (101), (004), (200), (211), (204) and (116) (JCDPS No. 21-1272), respectively [39,51]. While the hematite diffraction peaks appearing at  $2\theta=24.1^\circ, 33.1^\circ, 35.6^\circ, 40.8^\circ, 49.4^\circ, 54.1^\circ 62.4^\circ$  and  $64.0^\circ$  can be attributed to the (012), (104), (110),





(024), (116), (214) and (300) lattice plane of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (JCPDS 73-2234), respectively. It can be seen that Ar-plasma treatment modified the surface anatase TiO<sub>2</sub> to rutile phase with the amount correlating to exposure time, the result showed a new small peak of rutile phase TiO<sub>2</sub> at  $2\theta = 28.5^{\circ}$  corresponding to the (110) [48]. Also, XRD showed a modified photocatalytic surface, it was observed that a mixture of anatase and rutile phase were presented, which is in agreement with a previous study by Sun et al. [48]. Kawakami et al. [49] also reported a mix of anatase and rutile phase correlating to the Ar-plasma etching time.

The Ar-plasma treatment modifies the surface  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> by changing pure anatase TiO<sub>2</sub> to a mix of anatase and rutile phase, and this can be attributed to the energetic electrons in the plasma [48]. Also, since the lattice structure and ionic radius of Fe<sup>3+</sup> and Ti<sup>4+</sup> are similar, the iron could diffuse into the TiO<sub>2</sub> lattice converting the junction from anatase to rutile phase [48].

The average crystallite size of TiO<sub>2</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> nanocomposite and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> with various plasma treatment times ranged from 19.01  $\pm$  4.95–23.50  $\pm$  6.61 (Table 1). The average crystallite size was calculated from the XRD peak using the Debye-Scherrer equation as follows [57]:

$$d = \frac{\kappa \lambda}{\beta cos\theta} \tag{1}$$

where d is the crystallite size (nm),  $\lambda$  is the X-ray wavelength (nm), k is a constant related to the crystallite shape and normally taken as 0.89,  $\beta$  is the peak width at half maximum (FWHM) of the XRD peak and  $\theta$  is the diffraction angle [57]. The crystallite average size of  $TiO_2,\,\alpha\mbox{-}Fe_2O_3$  and  $\alpha$ -Fe<sub>2</sub>O<sub>2</sub>/TiO<sub>2</sub> nanocomposite was 19.01  $\pm$  4.95 nm, 21.78  $\pm$  1.74 nm and  $23.50 \pm 6.61$  nm, respectively. It can be seen that after plasma treatment of  $\alpha\text{-}Fe_2O_3/\text{Ti}O_2\text{-}5, \ \alpha\text{-}Fe_2O_3\text{Ti}O_2\text{-}10, \ \alpha\text{-}Fe_2O_3/\text{Ti}O_2\text{-}15$  and the crystalline size α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-20, reduced was from  $23.50 \pm 6.61$  nm to  $20.13 \pm 2.63$  nm. This may be attributed to effect of Ar plasma treatment on the surface of the sample. The smaller crystalline size resulted in a higher photocatalytic activity with a wider absorption of visible light irradiation. The crystallinity of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-untreated photocatalyst was 92.55%. After plasma modification on the surface, it can be seen that the crystallinity was 78.23-83.45% for the α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>, α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-5, α-Fe<sub>2</sub>O<sub>3</sub>TiO<sub>2</sub>-10,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-15 and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-20.

The UV–vis diffuse reflectance spectroscopy (DSR) absorption spectrum of TiO<sub>2</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> was recorded within the range of 200–800 nm, as shown in Fig. 4. It can be observed that TiO<sub>2</sub> shows the absorption of only UV light (420 nm), while  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> absorbed over the visible region (580 nm). It was observed that the heterojunction of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> photocatalyst shows wide absorption of UV–visible light. After plasma treatment, the absorption edge of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> shows significantly enhanced absorption odge of the plasma treated 400–700 nm. The stroger absorption edge of the plasma treated

#### Table 1

The average crystallite size, crystallinity, and band gap energy of various samples.

Sample	Average crystallite size (nm)	Crystallinity (%)	Band Gap Energy (eV)
TiO <sub>2</sub>	$19.01 \pm 4.95$	95.01	3.16
a-Fe <sub>2</sub> O <sub>3</sub>	$21.78 \pm 1.74$	65.82	2.02
α- Fe <sub>2</sub> O <sub>3</sub> /TiO <sub>2</sub>	$23.50 \pm 6.61$	92.55	2.18
a-Fe2O3/TiO2-	$21.76 \pm 7.82$	82.63	1.88
5			
a-Fe <sub>2</sub> O <sub>3</sub> /TiO <sub>2</sub> -	$21.50 \pm 3.76$	78.23	1.84
10			
a-Fe <sub>2</sub> O <sub>3</sub> /TiO <sub>2</sub> -	$21.74 \pm 5.31$	78.99	1.90
15			
a-Fe2O3/TiO2-	$20.13 \pm 2.63$	83.45	1.86
20			

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Fig. 4. UV–Vis absorption spectra of TiO<sub>2</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-15 and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-20.

 $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-5,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-15 and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-20 increases with increasing Ar-plasma treatment time of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>. The optical band gap energy of the TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles was calculated by Tac'c plot equation following [58]:

 $\alpha hv = A (hv - E_{\sigma})^{n/2}$ 

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where *a* is the absorbance, *h* is Planck's constant, *v* is the light frequency, *A* is a constant, *E<sub>g</sub>* is the band gap energy and the value of *n* depends on the type of optical transition [58]. Optical band gap energy of the sample is shown in Table 1 and Fig. 5. The TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> has band gap energy and was obtained at 3.16 eV and 2.02 eV, respectively. The band gap energy of photocatalyst  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> nanocomposite was found at 2.18 eV. Optical band gap energy of Ar-plasma treated photocatalyst were found to be 1.88, 1.84, 1.90 and 1.86 eV for  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-5,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-15 and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-20, respectively. From the above results, the band gap energy was reduced after plasma treatment from 2.18 eV to 1.84 eV. The reduced band gap energy shows Ar-plasma treatment. However, the optimum Ar plasma treatment time

was 10 min to produce a band gap of 1.84 eV ( $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10). Fig. 6 (a) shows the high-resolution XPS survey scan spectrum of the O 1s, Ti 2p and Fe 2p elements for both the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> untreated and Ar-plasma treated. The high resolution of O 1s is presented in Fig S1, the O 1s spectrum displayed a lattice structure centered at 528.84 eV, which was attributed to the lattice oxygen in the oxides O [43]. The weak peak at 531.27 eV was attributed to the defect oxygen vacancies (-OH), and the binding peak at 532.29 was attributed to the presence of surface oxygen, which is mainly attributed to lattice oxygen combined with Fe<sup>3</sup>-and Ti<sup>4+</sup> [39]. Hence, it can be seen that after plasma treatment with Ar [39]. Hence, it can be seen that after plasma treatment with Ar, contribution of the oxygen vacancies (OH) was increased with binding energy shifts from 532.29 eV to 531.80 eV. An additional binding peak at the binding energies of  $TiO_2$  corresponds to  $Ti^{3+}$  [40]. In addition, the high-resolution XPS of Fe is shown in Fig. 6 (b), the binding energies of Fe at 711.09 and 724.26 eV correspond to Fe  $2p_{3/2}$  and Fe  $2p_{1/2}$  of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, and satellite peak at 718.2 eV (Fe<sup>3+</sup>  $2p_{3/2}$ ) corresponds to the  $Fe^{3+}$  ion state in  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> [26,43]. The XPS result shows that plasma treatment of increased the Fe by 0.79%. The high-resolution scan spectrum of Ti in Fig. 6 (c) shows four binding peaks at 458.58, 464.10, 464.21 and 465.38 eV corresponding to  $Ti^{3+}$   $2p_{3/2}$ ,  $Ti^{4+}2p_{3/2}$ ,  $Ti^{3+}$  $2p_{1/2}$ , and  $T_1^{4+} 2p_{3/2}$ ,  $T_1^{2} 2p_{1/2}$  and  $T_1^{4+} 2p_{3/2}$ ,  $T_1^{2} 2p_{1/2}$ , and  $T_1^{4+} 2p_{3/2}$ , respectively. The split between the Ti  $2p_1^{3+}_{1/2}$  (5.9 eV), indicates the presence of  $T_1^{4+}$  in anatase  $TiO_2$ . The Ti  $2p_{1/2}$  peak, at 458.58 eV, is due to the formation of the Fe-O-Ti bond of nanocomposite in the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> [59]. All of the above results

(2)

 $E_{CB}$ 



Fig. 5. Tach Plot of (a) TiO<sub>2</sub>, (b) α-Fe<sub>2</sub>O<sub>3</sub>, α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>, (c) α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-5, α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10 and (d) α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-15, α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-20.

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indicate that the plasma modification on heterojunctions of  $\alpha\text{-}Fe_2O_3/\text{-}TiO_2$  nanocomposite have been synthesised successfully.

The properties and calculated valence bands (VB) of TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> were plotted from the XPS spectra of according to Bhoi et al. [60] and Naldoni et al. [61], as shown in Fig. 6 (d). The VB position of TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> were found to be 2.79 eV and 1.38 eV, respectively. Then the conduction band (CB) and band edge position of TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> were calculated using the equation as follows [60]:

$$a = E_{\rm VB} - E_{\rm g} \tag{3}$$

where  $E_{\rm CB}$  and  $E_{\rm VB}$  is the conduction band (CB) edge and valence band (VB) edge potential, respectively, and  $E_{\rm g}$  is the band gap energy of the semiconductor calculated from optical absorption. The calculated CB position of TiO\_2 and a-Fe\_2O\_3 was estimated to be - 0.27 eV and - 0.63 eV, respectively. The band position VB of  $\alpha$ -Fe\_2O\_3 is higher than the TiO\_2 band position. The CB band position  $\alpha$ -Fe\_2O\_3 is higher than that of TiO\_2 photocatalyst as shown in Table 2.

The PI spectra was used to investigate the recombination rate of the photogenerated electron hole-pair, presented in Fig. 7 (a). The PL spectra of TiO<sub>2</sub> nanoparticle had higher intensity, showing a lower separation efficiency of electron-hole pair. The emission intensity of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> nanocomposite was lower than that of pure TiO<sub>2</sub>, indicating reduced recombination rate of the photogenerated electron-hole pair. The Ar-plasma treated surface of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> nanocomposite had the lowest PL intensity, which means that the photogenerated charge carriers had high photocatalytic efficiency [62].

To confirm the charge transfer resistance, semicircle size in a Nyquist plot was investigated by EIS, shown in Fig. 7 (b). It was observed that

the Nyquist impedance plots for TiO<sub>2</sub> had a larger semicircle than  $\alpha$ -Fe<sub>2</sub>O<sub>2</sub>/TiO<sub>2</sub> or  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10, indicating a higher charge transfer resistance. After modification by plasma treatment,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10 had the smallest semicircle in the Nyquist impedance plot, suggesting that the charge transfer resistance was decreased [63]. The charge transfer resistance was 2.93, 2.24, and 1.90 kΩ for TiO<sub>2</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10, respectively. This result suggested that the plasma treatment of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> to and allowed for rapid migration between the charge carrier interface [47]. It was demonstrated that plasma modification of photocatalyst surface reduced the charge transfer resistance due to enhanced performance of photocatalytic activity.

The surface SEM image of the TiO<sub>2</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10 are shown in Fig. 8. The morphology of pure TiO<sub>2</sub> shows an average length of 5–7 µm and an average of diameter of 50–100 nm, as shown in Fig. 8 (a). It was observed in the SEM image that  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> shows a rough surface, Fig. 8 (b). While Fig. 8 (**c**-f) show that after Ar-plasma treatment, the spherical particles were slightly less densely packed on the surface with increasing Ar-plasma treatment time [43,64]. The conduct morphology of the photocatalytic activity of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> was characterized using TEM, HR-TEM and SAED, as shown in Fig. 9 (**a**-c). It can be seen in the TEM image of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>TiO<sub>2</sub> nanocomposite that the lattice spacing of 0.25 nm corresponds to (110) plane of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>. The selected area electron diffraction (SAED) pattern is shown in Fig. 9 (**d**), the SAED pattern formation shows (101), (004) and (200) for TiO<sub>2</sub> and (116), for  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, confirming the polycrystalline structure



Fig. 6. XPS survey spectra patterns of (a) α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10 XPS pattern (b) Fe2p, (c) Ti 2p, and (d) XPS valence band of α-Fe<sub>2</sub>O<sub>3</sub> and TiO<sub>2</sub>.

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Table 2

The band gap, calculated valence band (VB) and conduction band (CB) position of TiO\_2 and  $\alpha\text{-}Fe_2O_3.$ 

Sample	Band gap (eV)	E <sub>VB</sub> (eV)	E <sub>CB</sub> (eV)
TiO <sub>2</sub>	3.16	2.79	-0.27
α-Fe <sub>2</sub> O <sub>3</sub>	2.02	1.38	-0.63

and successful synthesis of  $\alpha\text{-}Fe_2O_3/\text{Ti}O_2$  nanocomposite, consistent with a report by Liu et al. [65].

The AFM was used to study the surface roughness of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>// TiO<sub>2</sub> heterojunction photocatalyst as shown in Fig S2 (**a**,**b**). It can be seen that the surface roughness of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>//TiO<sub>2</sub> photocatalyst is Ra = 46.71 nm and Ar-plasma treated is Ra = 44.87 nm. The AFM results revealed that plasma treatment of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10 decreased the surface roughness and increased homogeneity compared with untreated  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> photocatalyst nanocomposite [42].

The EDS mapping of plasma-treated heterojunction  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> photocatalytic nanocomposite confirmed typical deposited elements of O, Ti and Fe presented in Fig. S3 (a). The survey mapping of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/ TiO<sub>2</sub> after Ar plasma treatment shows the elements of O, Fe and Ti on the surface. It demonstrated that the surface was a uniform mix of both  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and TiO<sub>2</sub> after Ar-plasma treatment. After recycle for 5 times, the corresponding amounts of O, Ti, Fe, and C were 40.2, 32.9, 22.5 and 4.4 wt%, respectively (Fig S3 (b)).

#### 3.2. H<sub>2</sub> evolution

The photocatalytic H<sub>2</sub> evolution rates from different plasma treatment exposure times of heterojunction  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> nanocomposite are presented in Fig. 10 (a).  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> shows a photocatalytic H<sub>2</sub> evolution rate of 0.131 mmol h<sup>-1</sup>, which suggests a fast recombination rate of photogenerated electrons and holes. After surface plasma treatment of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>, the H<sub>2</sub> evolution rates increased to 0.84, 1.25, 1.05 and 0.920 mmol h<sup>-1</sup> for  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-5,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10 ranocomposite showed the highest H<sub>2</sub> evolution rate.

nanocomposite showed the highest H<sub>2</sub> evolution rate. The photocatalytic activities of surface plasma-treated  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> nanocomposite for long term H<sub>2</sub> evolution are presented in Fig. 10 (b), the photocatalytic activity of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> shows a low H<sub>2</sub> evolution rate. All plasma-treatment times resulted in increased photocatalytic activity of H<sub>2</sub> evolution compared to untreated  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> photocatalytic he highest H<sub>2</sub> evolution rate of plasma treated samples was 5.5 mmolh<sup>-1</sup> for 5 h reaction time.

The performance of photocatalytic recycling stability of Ar-plasma treated  $\alpha$ -Fe203/TiO2 for H2 evolution is presented in Fig. 10 (c), it can be seen that after the photocatalyst recycled five times, the sample shows high stability and good performance for H2 evolution. However, after a repeated cycling the stability slightly decreased as seen by a reduction of H2 evolution rate. The percent stability of H2 evolution rate was 94.27%, 93.30%, 92.18%, 90.75% and 89.32% for cycle 1, cycle 2, cycle 3, cycle 4 and cycle 5, respectively. Ibukun et al. [43] report a



Fig. 7. (a) PL spectra of photocatalyst and (b) Nyquist plot of  $TiO_2$ ,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10.



Fig. 8. SEM image of the surface of; (a) pure TiO<sub>2</sub>, (b) α-Fe<sub>2</sub>O<sub>3</sub> /TiO<sub>2</sub>, (c) α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-5 (d) α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10, (e) α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-15, and (f) α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-20.

similar percentage of reusability (94–96.7%) of plasma treated photocatalyst over four recycles. However, our result shows that there is no titanium detected. The iron was detected in the solution but there was not significantly difference. Statistical *t*-tests were conducted to compare the concentration of ferric ion before used and after recycle at each recycle time, the results indicate that it was not significantly of ferric ion concentration in solution after reaction at each cycle (p-value always < 0.05). Hence no leaching of ferric ion and titanium during photocatalytic reaction. It can be concluded from these results that the plasma treatment of the photocatalyst heterojunction could increase photocatalytic activity for H<sub>2</sub> evolution without contamination in aqueous solution during photoactivity.

The plasma treated photocatalyst exhibits a wide light response and demonstrates increased H<sub>2</sub> evolution rate over a longer lifetime of carrying charge. The intensity before and after recycling of the plasma treated photocatalyst was analyzed by XRD pattern, as shown in Fig. 10 (d). It can be seen that after fives cycles of reaction, the XRD peaks of the  $\alpha$ -fr<sub>2</sub>O<sub>2</sub>/TiO<sub>2</sub> photocatalyst did not significant change. Therefore, plasma treatment improved photocatalytic activity with high efficiency,

increased hydrogen evolution stability and reusability for H<sub>2</sub> evolution. In addition, Table 3 presents a comparison of H<sub>2</sub> evolution of various

photocatalyst materials that were treated with various plasma treatment methods from the literature. This shows that  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> has the highest H<sub>2</sub> evolution compared to previous reports.

The photocatalytic mechanism of H<sub>2</sub> evolution from water splitting for hydrogen evolution under visible light is shown in Fig. 11. Under visible light irradiation,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> an electron can excite from the valence band (VB) position to the conduction band (CB) of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> [26, 39]. The excited electron can then transfer to the CB in TiO<sub>2</sub>, because the CB band position of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> is higher than the TiO<sub>2</sub> band position, and the electrons accumulate at the lower-lying CB of TiO<sub>2</sub> [32,34]. This causes photogenerated holes to accumulate at the VB of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> which split a proton from a water molecule. The H<sup>+</sup> will react with an electron to generate a H atom and two combine to produce H<sub>2</sub> [27,39,69]. The photocatalytic reaction can be described by the following Eqs. (4)–(7) [27].

 $Ar \rightarrow Ar^{+} + e^{-} \rightarrow \alpha - Fe_2O_3/TiO_2 \xrightarrow{h\nu} TiO_2 - Fe_2O_3(e_{cb}^{-} + h_{vb}^{+})$ (4)



Fig. 9. TEM image of plasma treated  $\alpha\mbox{-}Fe_2O_3/TiO_2\mbox{-}10$  (a,b) HR-TEM (c) and (d) SAED pattern.

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$2e^{cb}{+}2H^+{\rightarrow}H^+{+}H^-{\rightarrow}H_2$	(5)
$(h_{vh}^+)$ +H <sub>2</sub> O $\rightarrow$ H <sub>2</sub> O <sup>+</sup> $\rightarrow$ OH + H <sup>+</sup>	(6)

A OUL AUT : O	
$2 \text{ OH} \rightarrow 2 \text{H}^{-} + \text{O}_{2}$	(7)

### 4. Conclusions

The photocatalytic heterojunction of  $\alpha\mbox{-Fe}_2O_3/TiO_2$  nanocomposite was successfully synthesized by the impregnation method and modified further using Ar-plasma for photocatalytic  $H_2$  evolution.

 Table 3

 Comparison of plasma treatment on photocatalytic materials for  $H_2$  evolution from the literature.

Catalyst	Plasma	Condition	H <sub>2</sub> evolution	Ref.
α-Fe <sub>2</sub> O <sub>3</sub> / TiO <sub>2</sub>	Ar-plasma	10 min, 50 W	1250 µmol h <sup>-1</sup>	This work
TiO <sub>2</sub> /GO	Ar-plasma	30 min, 400 Pa	170 µmol h <sup>-1</sup>	[45]
Pt/g-C <sub>3</sub> N <sub>4</sub>	Ar-Plasma	30 min, 1400 W	496.7 μmol h <sup>-1</sup> g <sup>-</sup>	[47]
Ni/NiO/ TiO <sub>2</sub>	NH <sub>3</sub> - Plasma	3–20 min, 10 kHz	$1210 \ \mu mol \ h^{-1}$	[66]
Ti <sub>3</sub> C <sub>2</sub> T <sub>x</sub> /Cd <sub>5</sub>	$O_2/N_2$	30 min,1400 W	825 µmol h <sup>-1</sup> g <sup>-1</sup>	[67]
NiO/Ta <sub>2</sub> O <sub>5</sub>	Ar-Plasma	30 min/NA	299 µmol h <sup>-1</sup>	[68]
NiO/ZrO2	Ar-Plasma	30 min/NA	172 µmol h-1	[68]



Fig. 10. Comparison of H<sub>2</sub> evolution rates (a) hydrogen rate of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-5,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-15, and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-20, (b) photocatalytic long term H<sub>2</sub> evolution of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>, Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-5,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-15, and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-20, (c) recycling long term H<sub>2</sub> evolution of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10 plasma and (d) XRD pattern before and after recycling.

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Fig. 11. Mechanism of photocatalysis over  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> under visible light.

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- > The morphology of Ar-plasma modified α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> nanocomposite changed from anatase TiO2 to mixed anatase and rutile phase. The average crystallite size decreased from 23.50 to 20.13 nm and the band gap energy reduced from 1.80 to 1.93 eV.
- Ar-plasma modification of surface showed defect oxygen vacancies and Ti<sup>3+</sup> in the interaction between Fe<sup>3+</sup> and Ti<sup>4+</sup>. The result demonstrated improved photocatalytic activity and good stability for a longer lifetime, with a reduced recombination rate of the photo-generated electron-hole pair.
- > The plasma treatment time of 10 min showed the highest photocatalytic activity of  $H_2$  evolution, which was 1.25 mmol h<sup>-1</sup> for  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10 nanocomposite, high efficiencies, good stability, and good reusability for H2 evolution.

### CRediT authorship contribution statement

Weerapong Bootluck: Investigation, Visualization and Original draft preparation. Thawat Chittrakarn: Writing - review & editing. Kuaanan Techato: Writing - review & editing. Watsa Khongnakorn: Conceptualization, Methodology, Visualization, Validation, Writing original draft preparation, Supervision and Writing - review & editing.

### **Declaration of Competing Interest**

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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#### Appendix A. Supporting information

Supplementary data associated with this article can be found in the online version at doi:10.1016/j.jece.2021.105660.

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# CHAPTER 4 CONCLUSIONS

In summary, the development of photocatalytic material of improving photocatalytic activity by noble metal deposition and plasma treatment for hydrogen heterojunction production. The photocatalyst of S-scheme  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> nanocomposite with the ternary of Pd were synthesis by sol-gel methods for increased H<sub>2</sub> production rate. As the result of the characterization, the average crystallite size was 22 nm. The band gap energy was decreased from 2.18 to 2.00 eV and the stronger intensity absorption under visible light. The deposition of Pd ternary over  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> was improve the recombination and separation efficiency of charge carriers. The H<sub>2</sub> evolution rate was increased from 74.60 µmol h<sup>-1</sup> to 349.54 µmol h<sup>-1</sup> and the reusability was decreased 10 % under visible light. The possible mechanism of heterojunction photocatalytic has been presented in S-Scheme heterojunction and Type-II heterojunction has been descript in Chapter 2 and Chapter 3.



Fig. 4.1 Mechanism of photocatalyst over  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> and Pd cocatalyst in S-Scheme and Type-II heterojunction

Table 4.1 Summarize of characterization and	performance of p	photocatalyst	samples
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Photocatalyst	UV-vis (λ,nm)	Average crystallite size (nm)	Crystallinity (%)	Band gap Energy (eV)	$H_2$ production (µmol h <sup>-1</sup> )	Reusability
$\alpha$ -Fe <sub>2</sub> O <sub>3</sub> /TiO <sub>2</sub>	≥420	23.5	92.55	2.18	74.60	NA
α-Fe <sub>2</sub> O <sub>3</sub> /TiO <sub>2</sub> -Pd	≥420	22.11	94.54	2.00	349.54	90%
α-Fe <sub>2</sub> O <sub>3</sub> /TiO <sub>2</sub> -10	≥420	21.50	78.23	1.84	1250	94%

The modification of photocatalytic heterojunction  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> by Ar plasma at 50 watts of 5-20 min was investigated. As a result, the plasma modification was obtained mixed of anatase and rutile phase were presented, the average crystal size was decreased from 25 nm to 20 nm and the band gap energy was reduced from 2.18 eV to 1.84 eV. The charge transfer resistance was reduced from 2.24 to 1.90 k $\Omega$  for  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10. The Ar plasma treatment at 10 min ( $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10) shows the highest H<sub>2</sub> production rate at 1250 µmol h<sup>-1</sup> when compare with the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>. Therefore, the concludes in this work to the development of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> photocatalyst of characterization and the performance is summarized in Table 4.1 Chapter 4.

## **APPENDIX**

## PAPER I SUPPORTING INFORMATION

# **Supporting Information**

# S-scheme $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> photocatalyst with Pd cocatalyst for enhanced photocatalytic H<sub>2</sub> production activity and stability

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Fig. S1: XPS of O 1s spectra of  $\alpha$ - Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-Pd



Fig. S2: Mott Schottky plot for  $TiO_2,\,\alpha\mbox{-}Fe_2O_3,\,\alpha\mbox{-}Fe_2O_3/TiO_2$  and  $\alpha\mbox{-}Fe_2O_3/TiO_2\mbox{-}Pd$ 

## PAPER II SUPPORTING INFORMATION

# **Supporting Information**

Modification of surface α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> photocatalyst nanocomposite with enhanced photocatalytic activity by Ar gas plasma treatment for hydrogen evolution

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Fig. S1. XPS pattern of O1s  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10.



Fig. S2. AFM of (a)  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> and (b)  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10.



**Fig. S3.** EDS mapping spectrum of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10, (a) before used and (b) after used 5 times.



Fig. S4. SEM image of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub>-10, (a) before used and (b) after used 5 times.

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- The research grant from Graduate School, Prince of Songkla University
- The scholarship from the Center of Excellence in Membrane science and Technology, Prince of Songkla University

## Publication

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- **Bootluck, W**., Chittrakarn, T., Techato, K., and Khongnakorn, W. 2021. Modification of surface  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> photocatalyst nanocomposite with enhanced photocatalytic activity by Ar gas plasma treatment for hydrogen evolution. Journal of Environmental Chemical Engineering, 9 (4), 105660.

### **International Conference**

- Weerapong Bootluck, Thawat Chittrakarn, Kuaanan Techato, Watsa Khongnakorn. Modification of surface α-Fe<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> photocatalyst nanocomposite with enhanced photocatalytic activity by Ar gas plasma treatment for hydrogen evolution. Regional Congress on Membrane Technology 2020 (RCOM 2020) and Regional Conference on Environmental Engineering (RCEnvE 2020). January 16-17 Johor, Malaysia, 16-17, 2021
- W. Bootluck, N. Junchoob, N. Meesiang, W. Khongnakorn and T. Chittrakarn. Dye Sensitizer of Chlorophyll Mixed with TiO<sub>2</sub> nanoparticle in photocatalytic for enhancing the hydrogen production. International Conference on Membrane Science and Technology (MST2019). Nanyang Technology University, Singapore, June 13-14, 2019
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